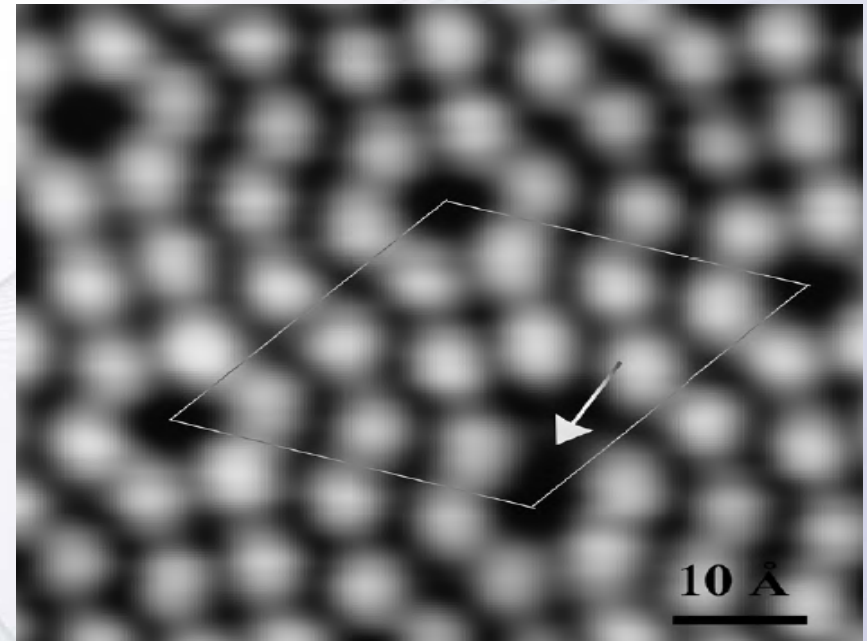


Cacat Kristal



Oleh

Dr. H. Harun Nasrudin, M.S.

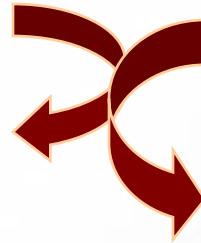


Samik, S.Si., M.Si.







OVERVIEW

PENGERTIAN



Classification of Defects

-  **Point defects**: solute atoms (strength, conductivity)
-  **Line defects**: dislocations (plastic deformation)
-  **Surface defects**: external surface (crystal shape)
-  **Volume defects**: voids, inclusions (fracture)



PENGERTIAN

An ideal crystal can be described in terms a three-dimensionally periodic arrangement of points called lattice and an atom or group of atoms associated with each lattice point called motif:

$$\text{Crystal} = \text{Lattice} + \text{Motif}$$

However, there can be deviations from this ideality.



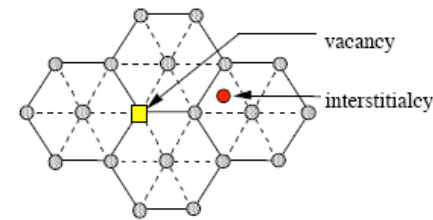
These deviations are known as **CRYSTAL DEFECTS**.

POINT DEFECTS



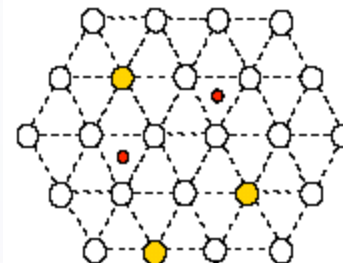
Intrinsik defects : Occur in pure substances

- @ Schottky defects
- @ Frenkel defects



Extrinsik defects Due to impurities

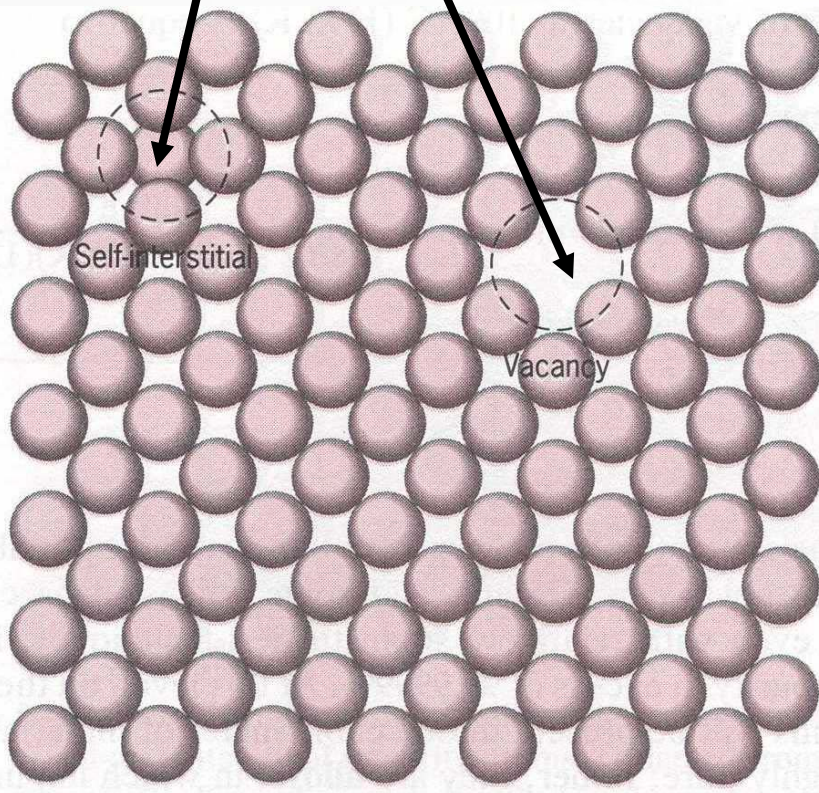
- @ Substitutional solid solutions
- @ Interstitial solid solutions



Point Defects

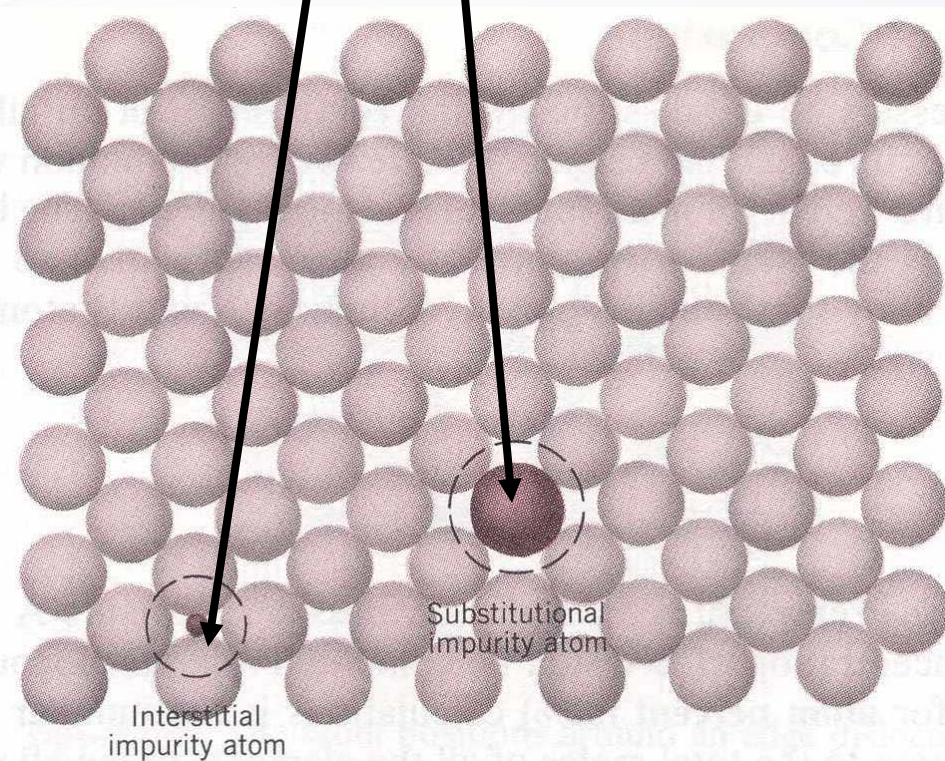
– *Intrinsic* defects

- *Vacancy*
- *Self-interstitial*



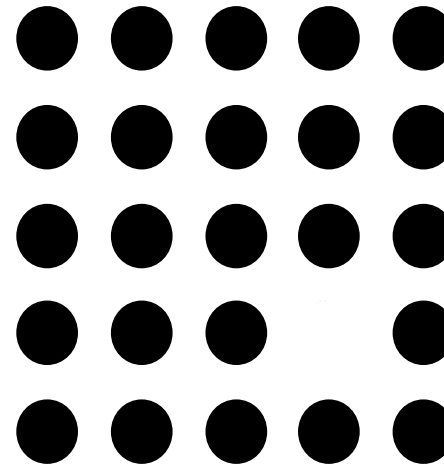
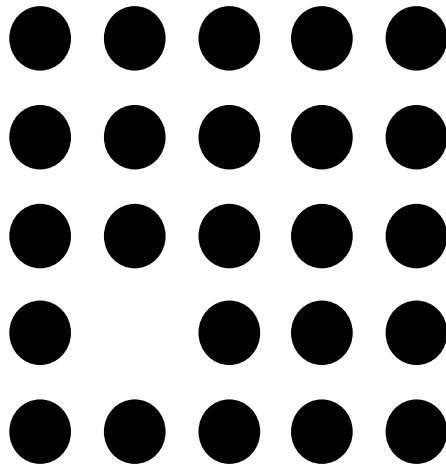
– *Extrinsic* defects

- *Substitutional impurity*
- *Interstitial impurity*



Vacancy

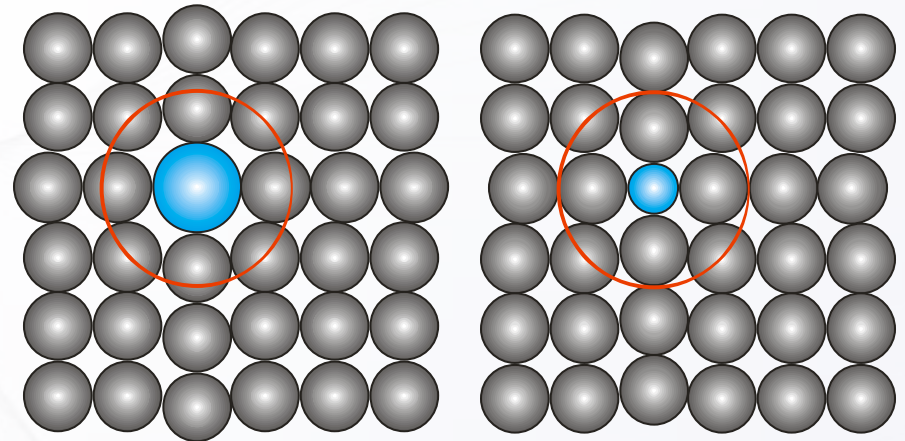
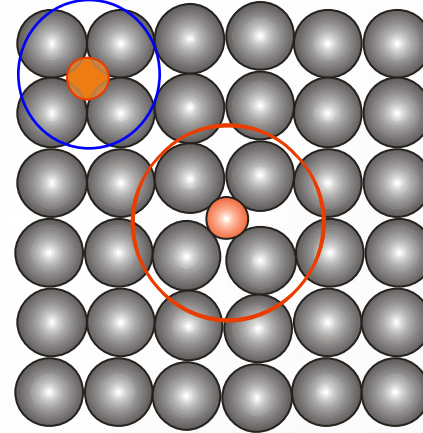
- ❑ Missing atom from an atomic site
- ❑ Atoms around the vacancy displaced
- ❑ Tensile stress field produced in the vicinity



Impurity

Interstitial

Substitutional



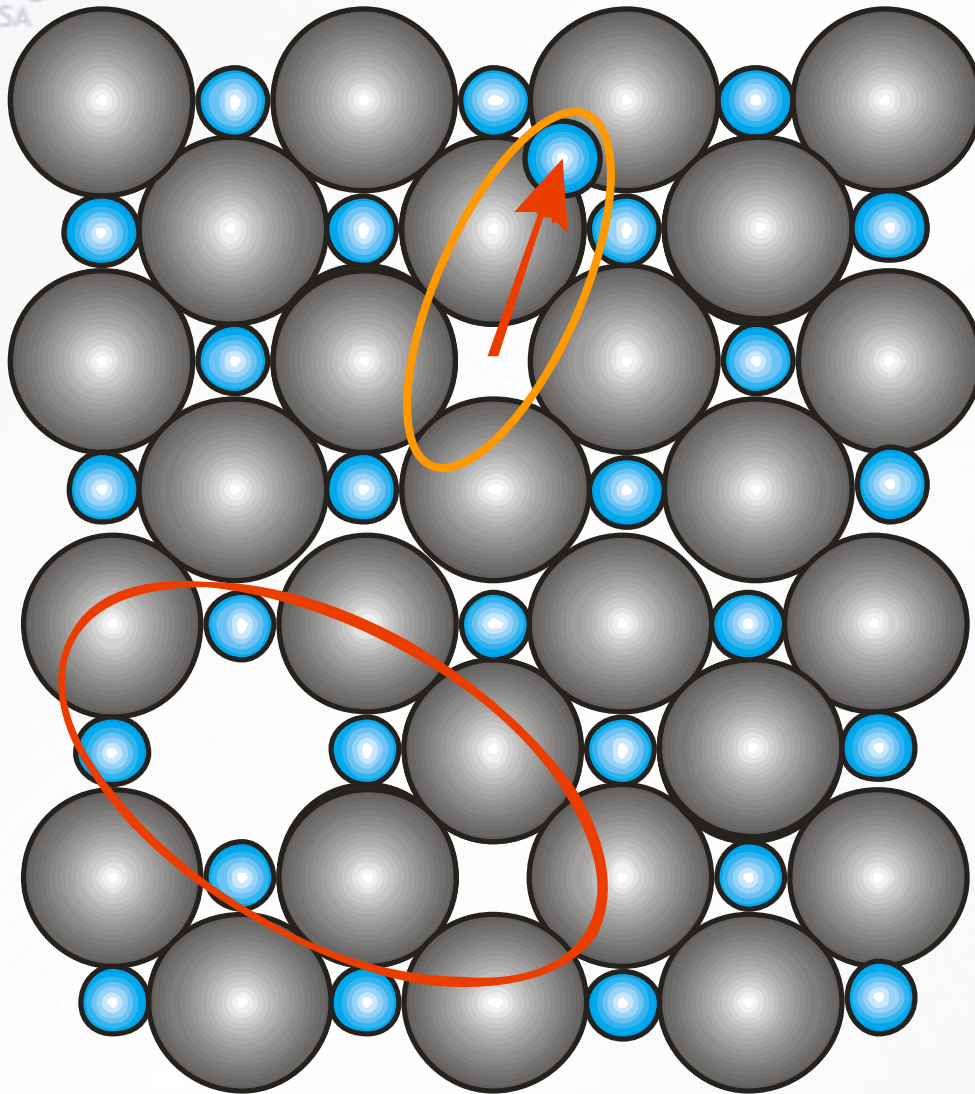
❑ SUBSTITUTIONAL IMPURITY

- Foreign atom replacing the parent atom in the crystal
- E.g. **Cu** sitting in the lattice site of FCC-**Ni**

❑ INTERSTITIAL IMPURITY

- Foreign atom sitting in the void of a crystal
- E.g. **C** sitting in the octahedral void in HT FCC-**Fe**

Defects in ionic solids



Frenkel defect

Cation vacancy
+
cation interstitial

Schottky defect

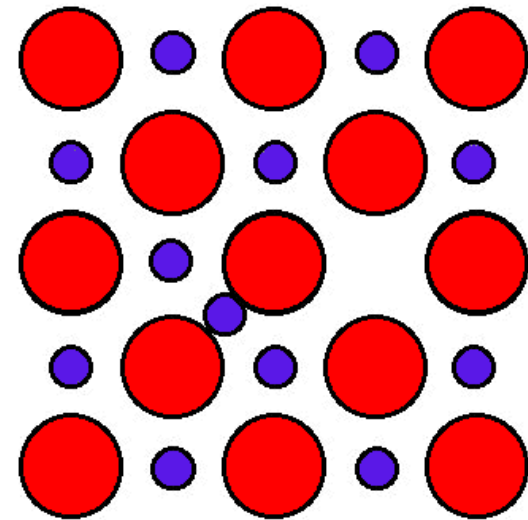
Cation vacancy
+
anion vacancy

Intrinsik Defects -Frenkel

Often a vacancy and interstitial occur together - an ion is displaced from its site into an interstitial position.

This is a **Frenkel Defect** (common in e.g. AgCl) and charge balance is maintained.

Frenkel defects can be induced by irradiation of a sample



Frenkel Defect

Extrinsic defects (due to impurities)

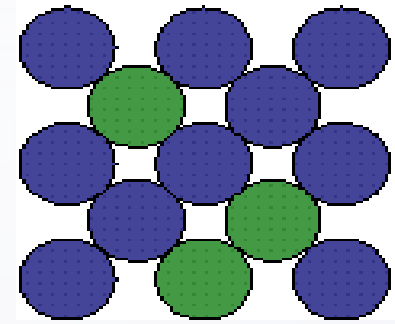
Impurities or **dopants** in a solid are any atom(s) of a type that do not belong in the perfect crystal structure (see 'extrinsic semiconductors')

The host crystal with impurities is called a **solid solution**

Substitutional solid solutions

Impurity atoms occupy the same sites of the **host atoms**

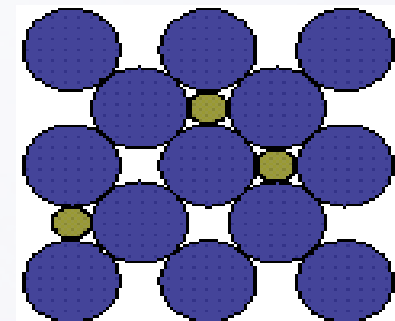
Impurities "substitute" for the host atoms



Interstitial solid solutions

Impurity atoms occupy interstices in the **host crystal** structure

Impurities usually have a small size compared to the host atoms



Impurity defects

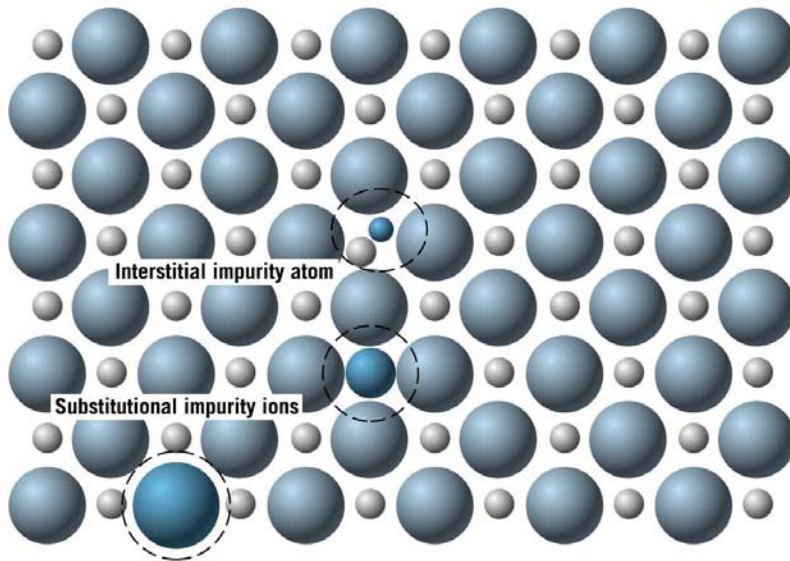


FIGURE 12.23 Schematic representations of interstitial, anion-substitutional, and cation-substitutional impurity atoms in an ionic compound. (Adapted from W. G. Moffatt, G. W. Pearsall, and J. Wulff, *The Structure and Properties of Materials*, Vol. 1, *Structure*, p. 78. Copyright © 1964 by John Wiley & Sons, New York. Reprinted by permission of John Wiley & Sons, Inc.)

Charge neutrality must be maintained.

Thus, if a substitutional impurity has a different charge than the substituted ion, another defect (or defects) must be present to balance it out.

Non-stoichiometry often results.

Cationic

Ca instead of Na in NaCl

B instead of Si in SiO_2

Anionic

O instead of Cl in NaCl

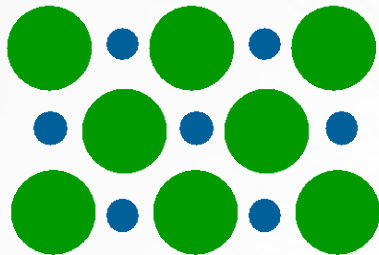
O instead of N in GaN

IMPURITIES

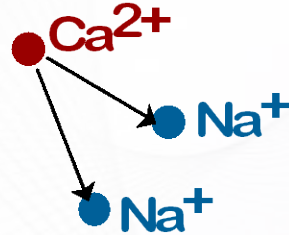
- Impurities must satisfy **charge balance**



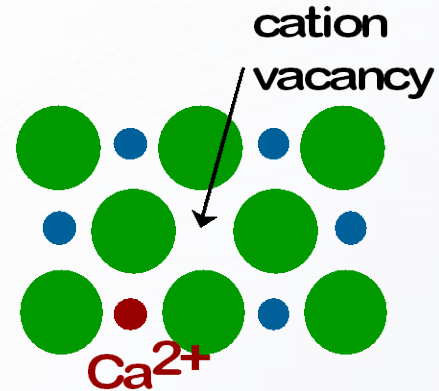
- Substitutional cation impurity



initial geometry

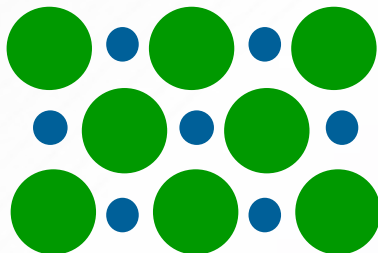


Ca^{2+} impurity

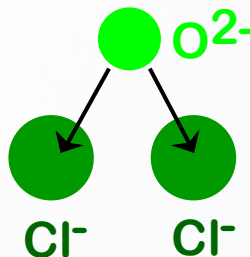


resulting geometry

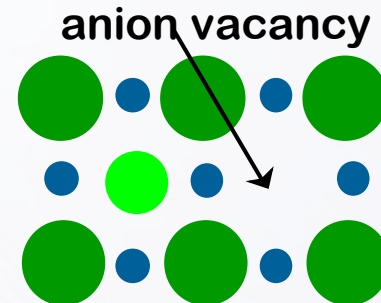
- Substitutional anion impurity



initial geometry



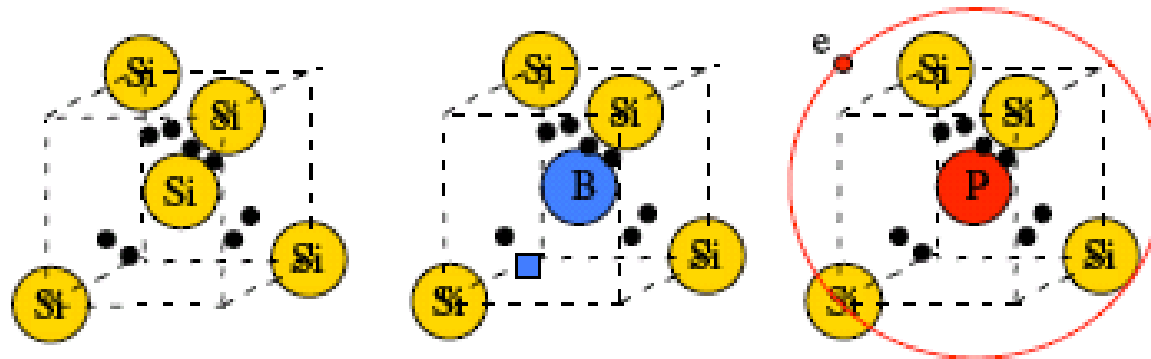
O^{2-} impurity



resulting geometry

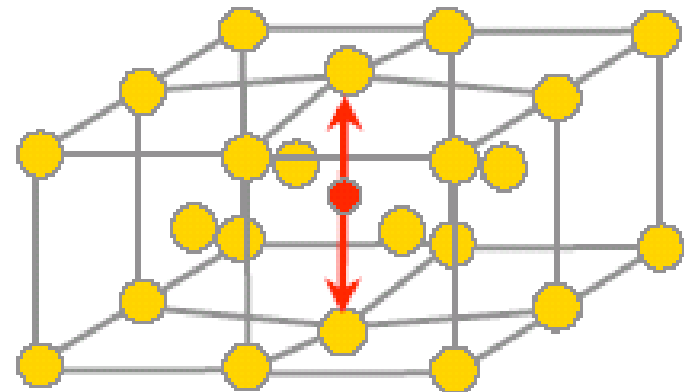
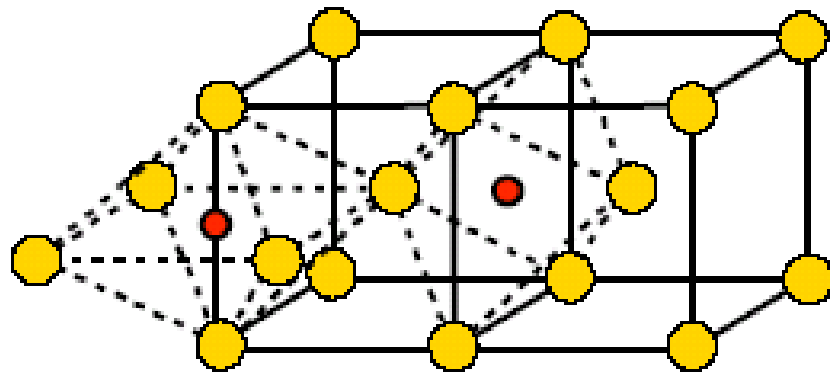
Substitutional Solutes:

Electrical Defects in Semiconductors



- Solutes can control the number and type of carriers in Si
- P ($z = 5$) introduces an electron in an excited state
 - Electron can be liberated to conduct electricity
 - “Donor” solutes create “n-type” extrinsic semiconductors
- B ($z = 3$) leaves a hole in a bonding state
 - Hole can “accept” an electron to create a mobile positive charge
 - “Acceptor” solutes create “p-type” extrinsic semiconductors

Interstitial Solutes: Carbon in Steel



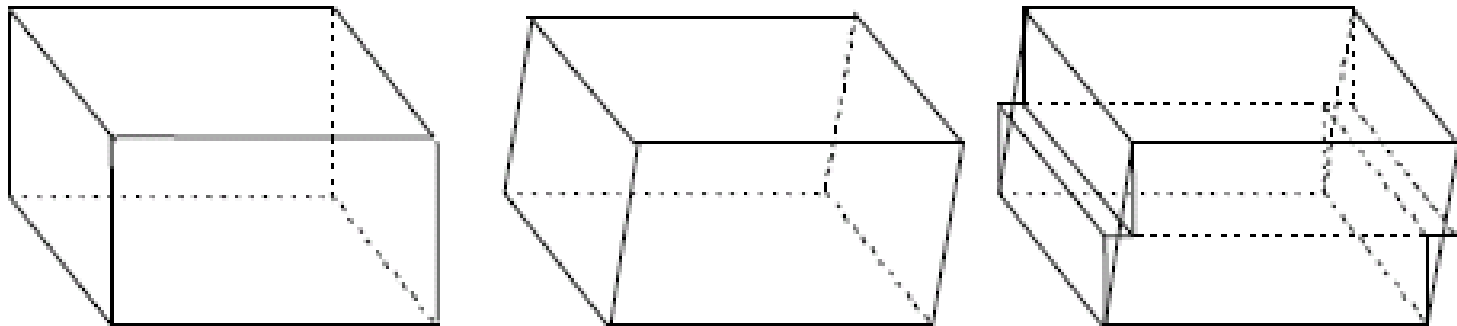
- Carbon interstitials fill octahedral sites in bcc Fe
 - Octahedral has larger volume (smaller radius) than tetrahedral
- Octahedral sites are asymmetric
 - Sites distort, creating local strain
 - Makes deformation difficult, strengthens material

LINE DEFECTS (DISLOCATIONS)

- @ Plastic deformations by Slip
- @ Edge Dislocations
- @ Screw Dislocations
- @ Mixed Dislocation

Line Defects:

Plastic Deformation by Slip



- Plastic deformation is a change in shape
 - Changes in shape happen by shear
 - Equivalent to simple shear shown
- Shear happens by “slip”
 - Planes of atoms slip over one another like cards in a deck
 - Slip is accomplished by linear defects - “dislocations”

Mechanism of plastic deformation in crystals: dislocation glide, or slip of atomic planes (atomic planes move one by one via the formation and movement of dislocations, rather than all the planes move simultaneously)

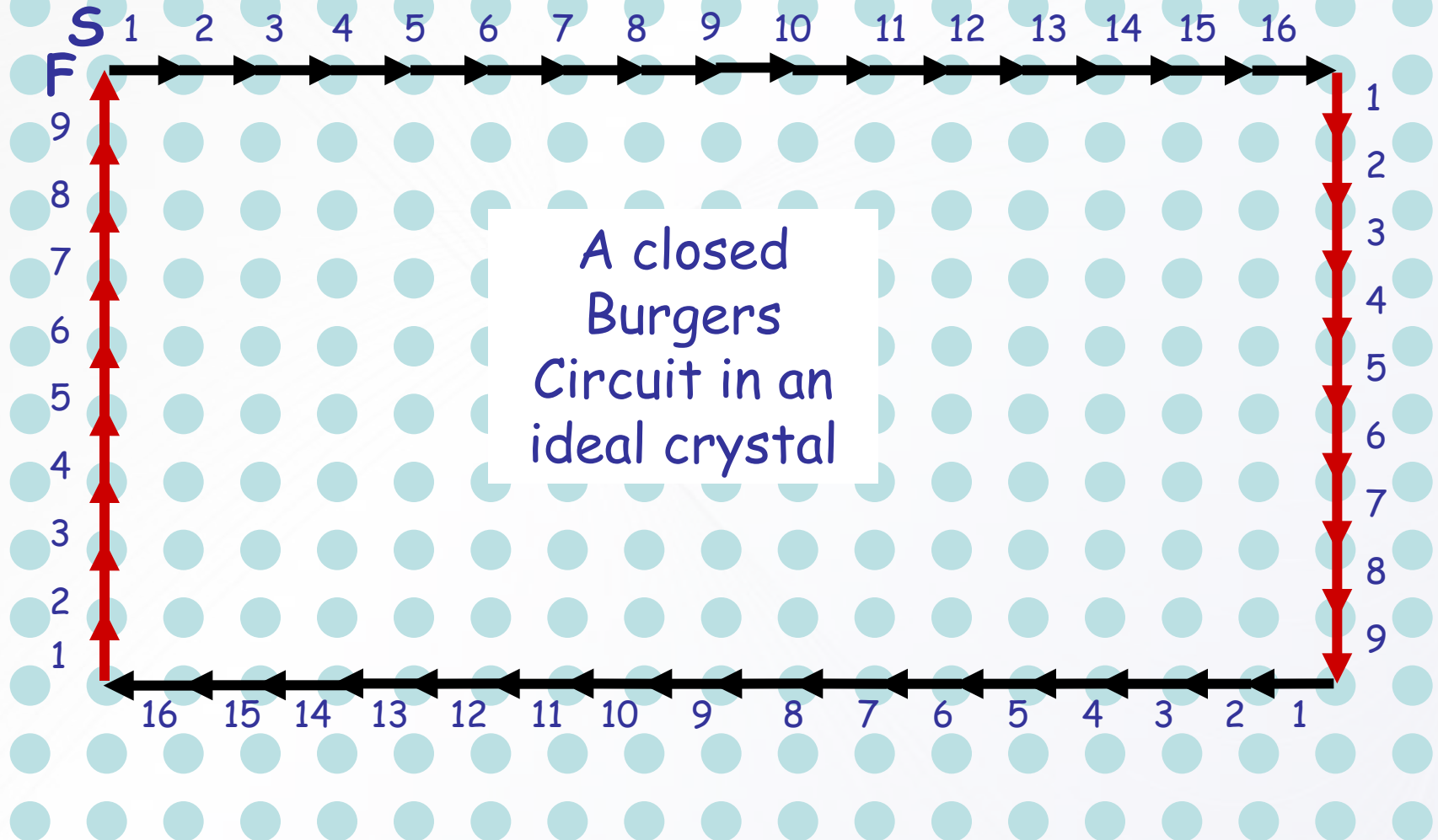
Burgers vector

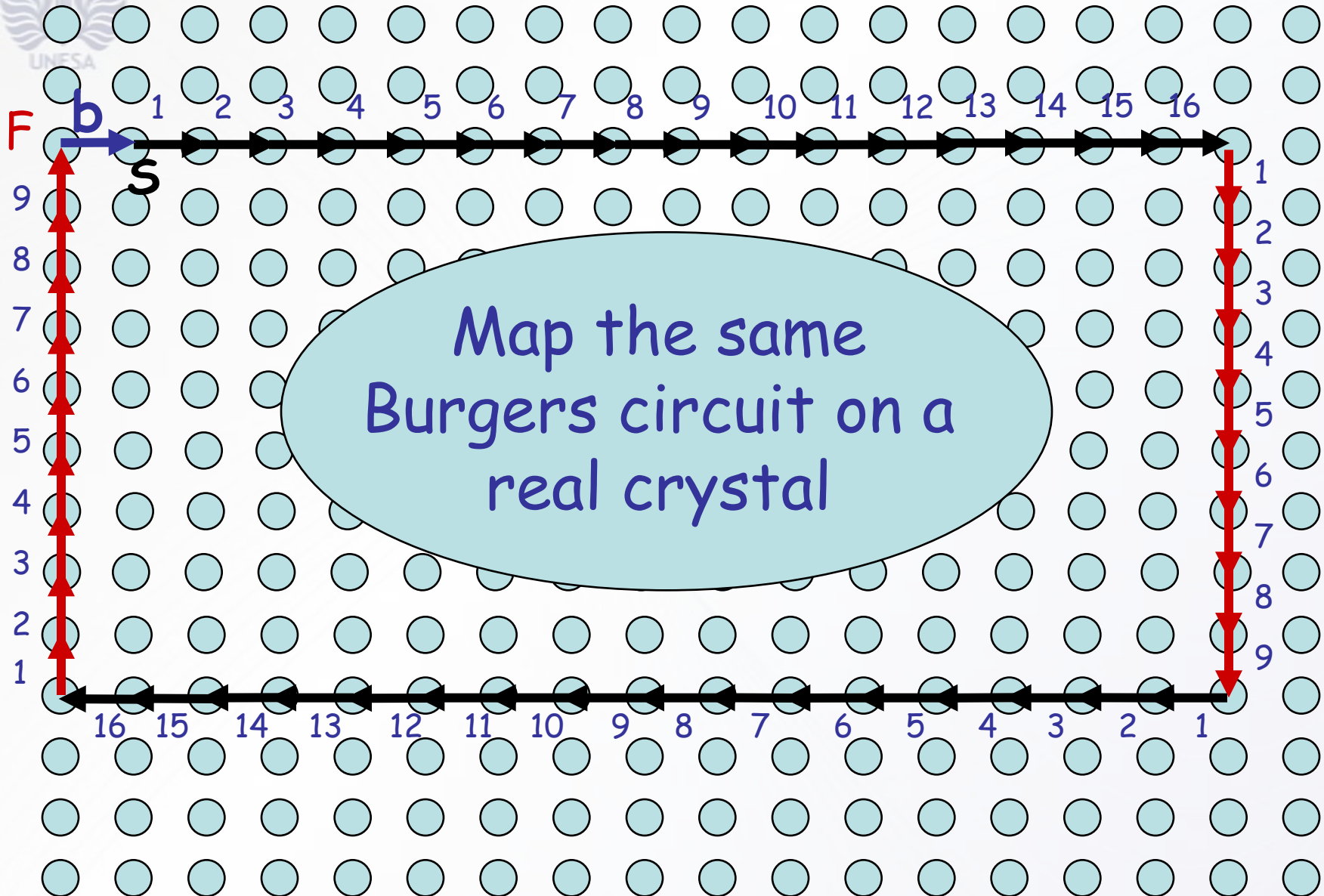


Johannes Martinus
BURGERS

~~Burger's~~ vector

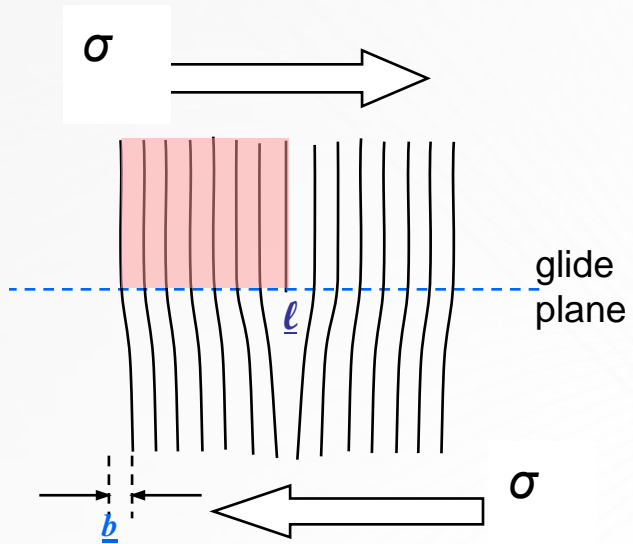
 Burgers vector





RHFS convention

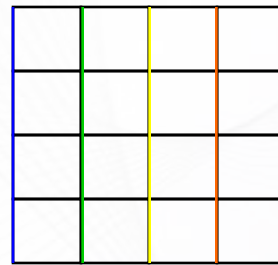
Definitions



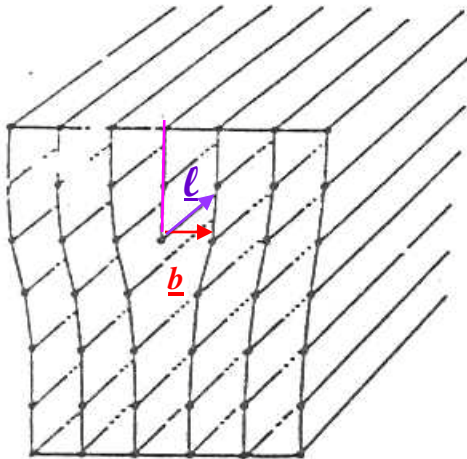
- The boundary plane across which shear occurs is the **glide plane**
- The boundary line that separates slipped (**red**) and unslipped regions is the **dislocation line** or **axis ℓ**
- The direction and magnitude of slip = **Burger's vector, \underline{b}**
- \underline{b} is in general a lattice vector, so there is no long range mismatch between slipped and unslipped planes
- If \underline{b} is parallel to ℓ dislocation is '**screw**'
- If \underline{b} is perpendicular to ℓ dislocation is '**edge**'

Edge and Screw Dislocations

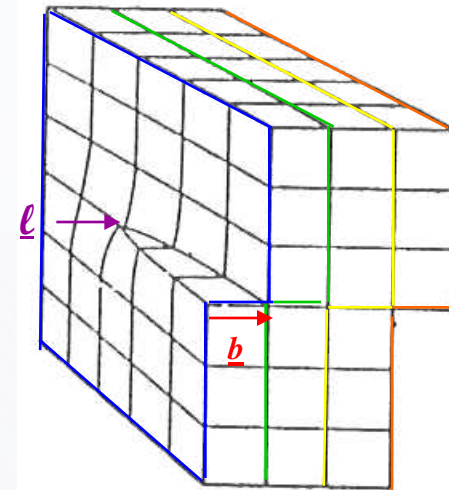
Perfect crystal lattice

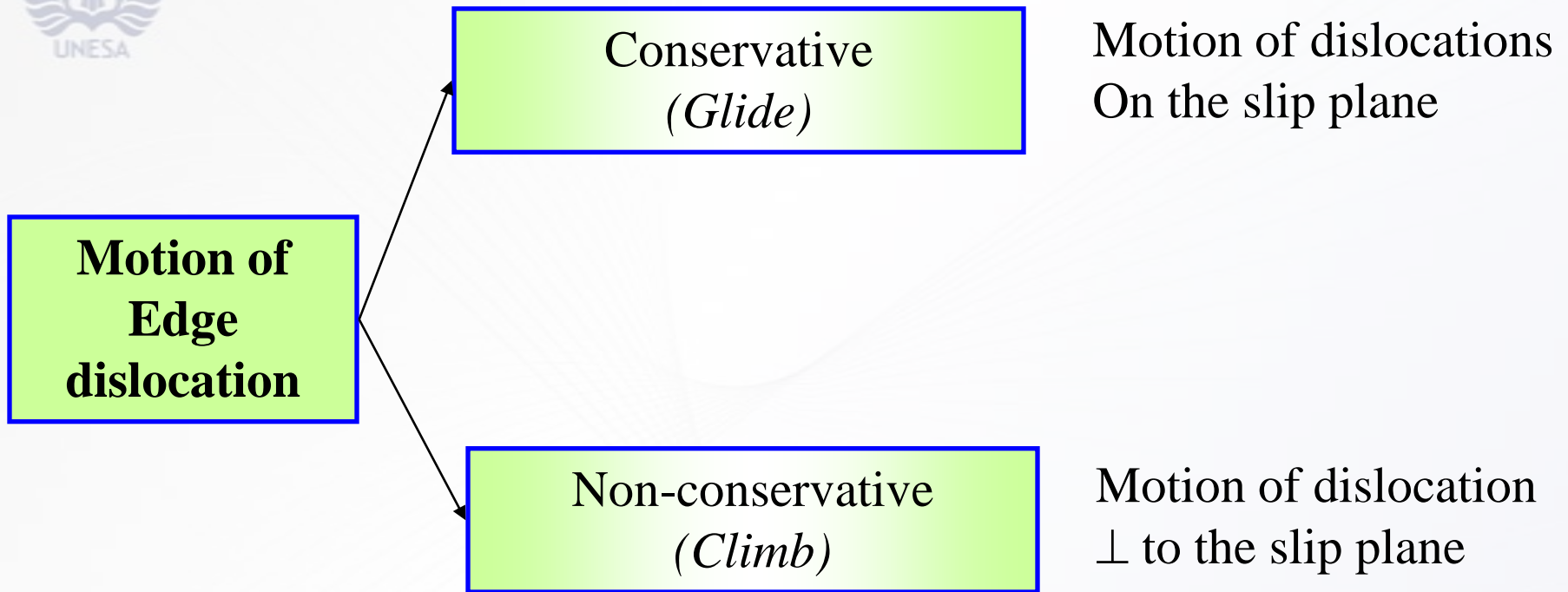


Edge dislocation:
“extra plane”



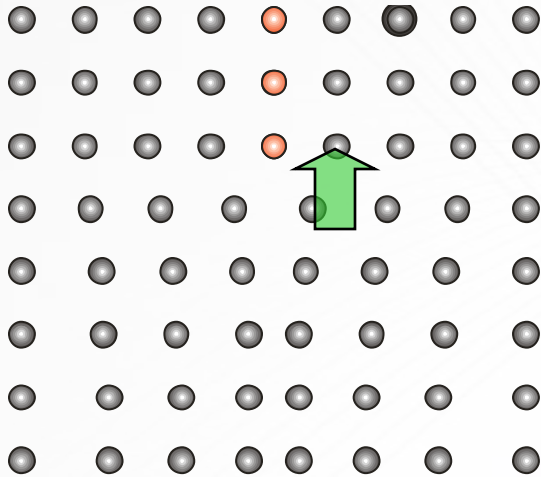
Screw dislocation
(distortion of the
crystal)





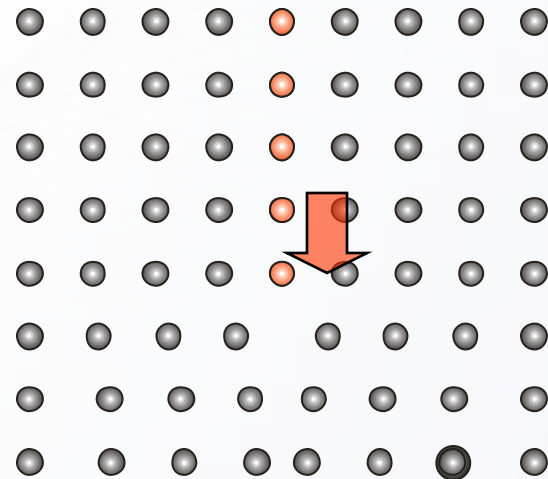
- ❑ For edge dislocation: as $\mathbf{b} \perp \mathbf{l}$
- ❑ Climb involves addition or subtraction of a row of atoms below the half plane
 - positive climb = climb up \rightarrow removal of a plane of atoms
 - negative climb = climb down \rightarrow addition of a plane of atoms

Edge Climb



Positive climb

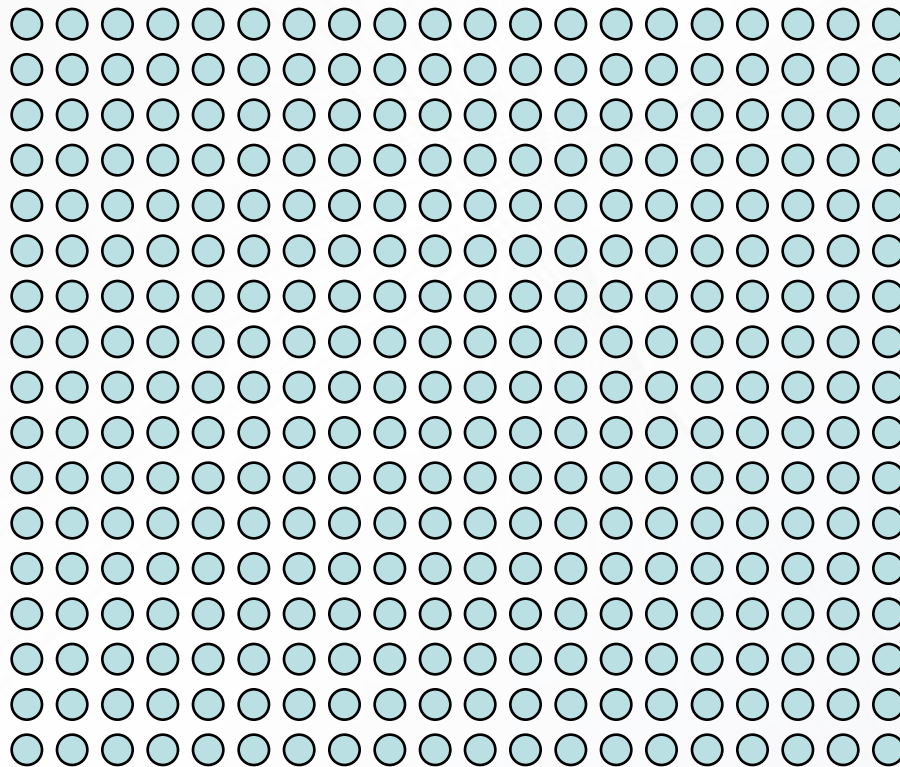
Removal of a row of atoms



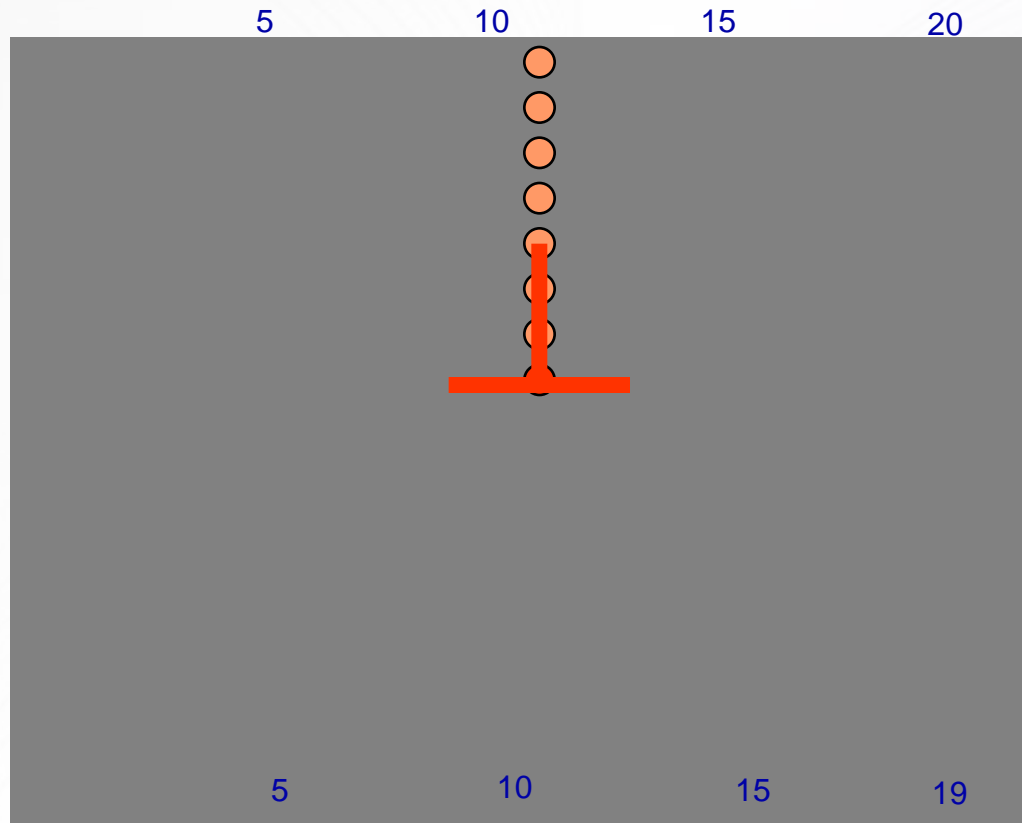
Negative climb

Addition of a row of atoms

Let's look at the atoms in a perfect crystal

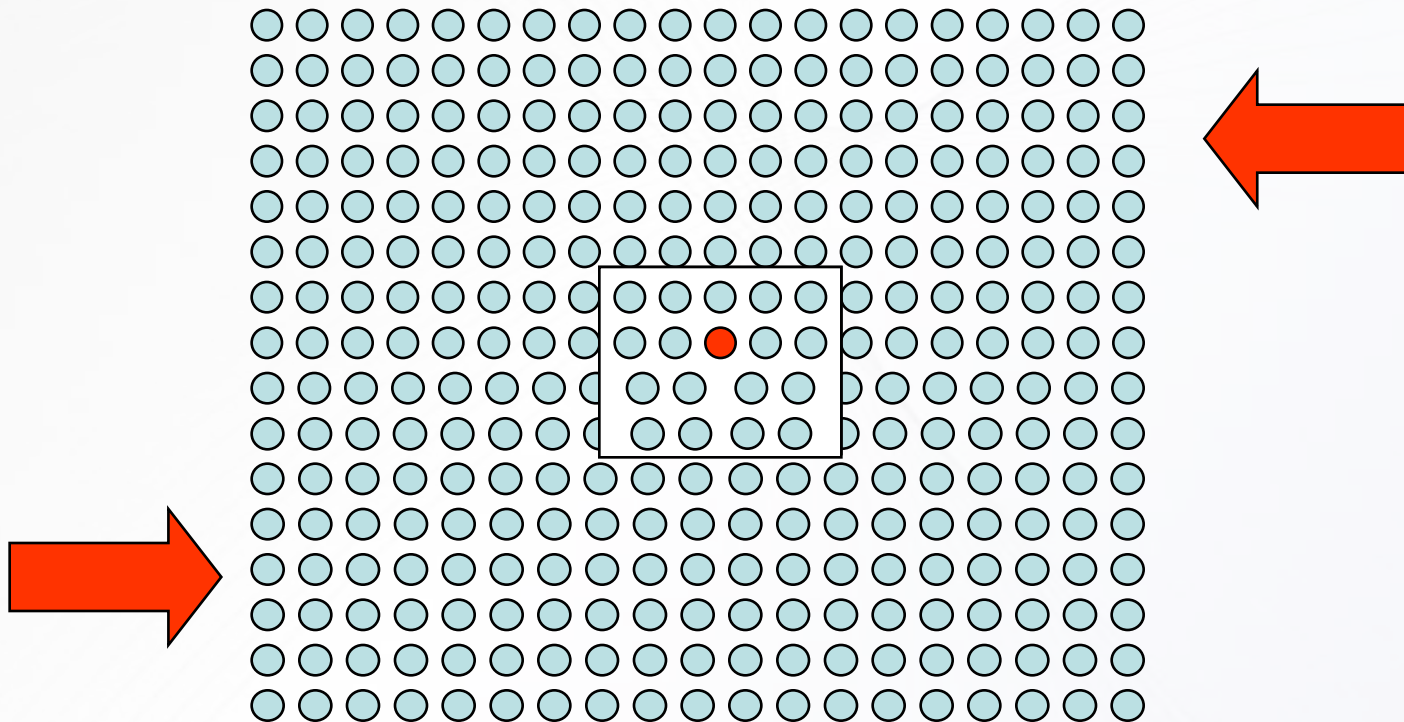


Let's look at the atoms in a realistic crystal

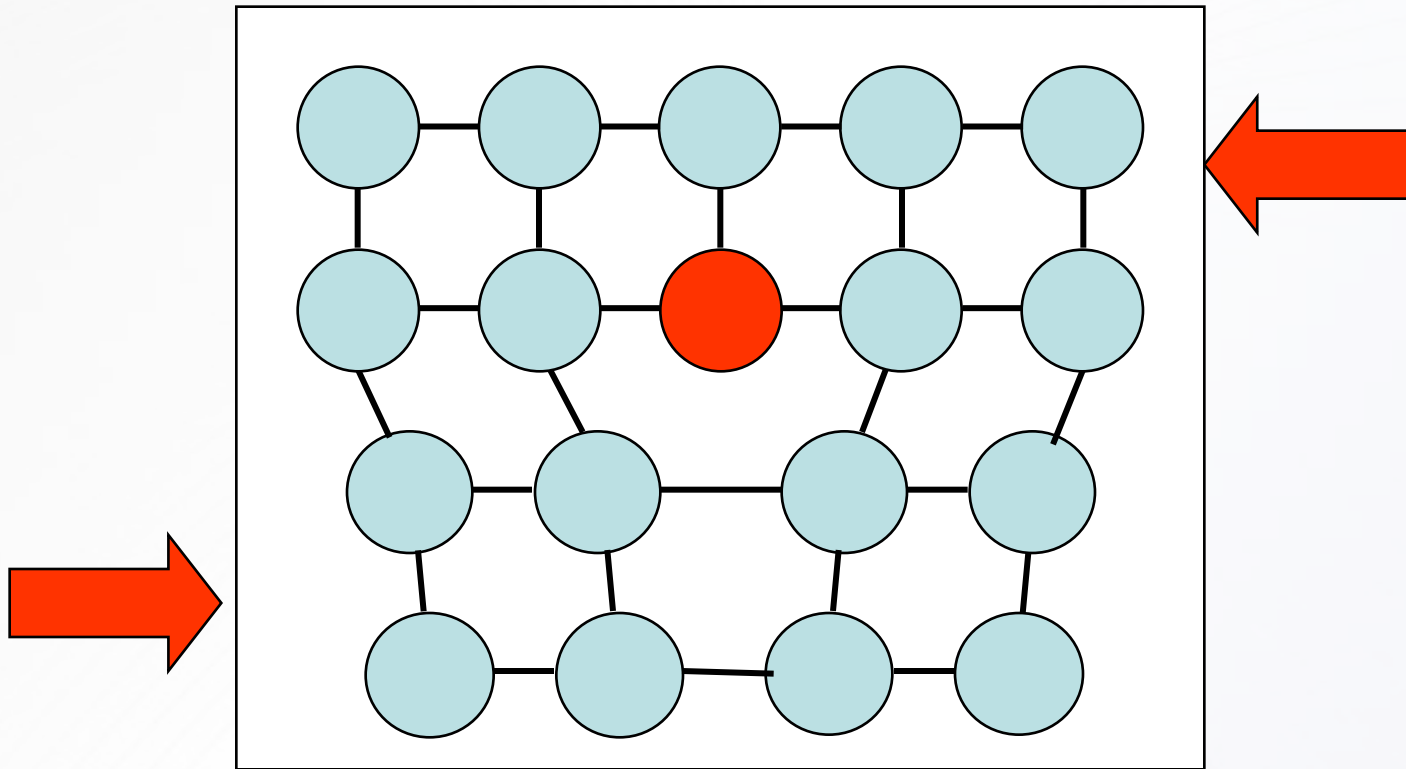


⊥ = “Edge dislocation”

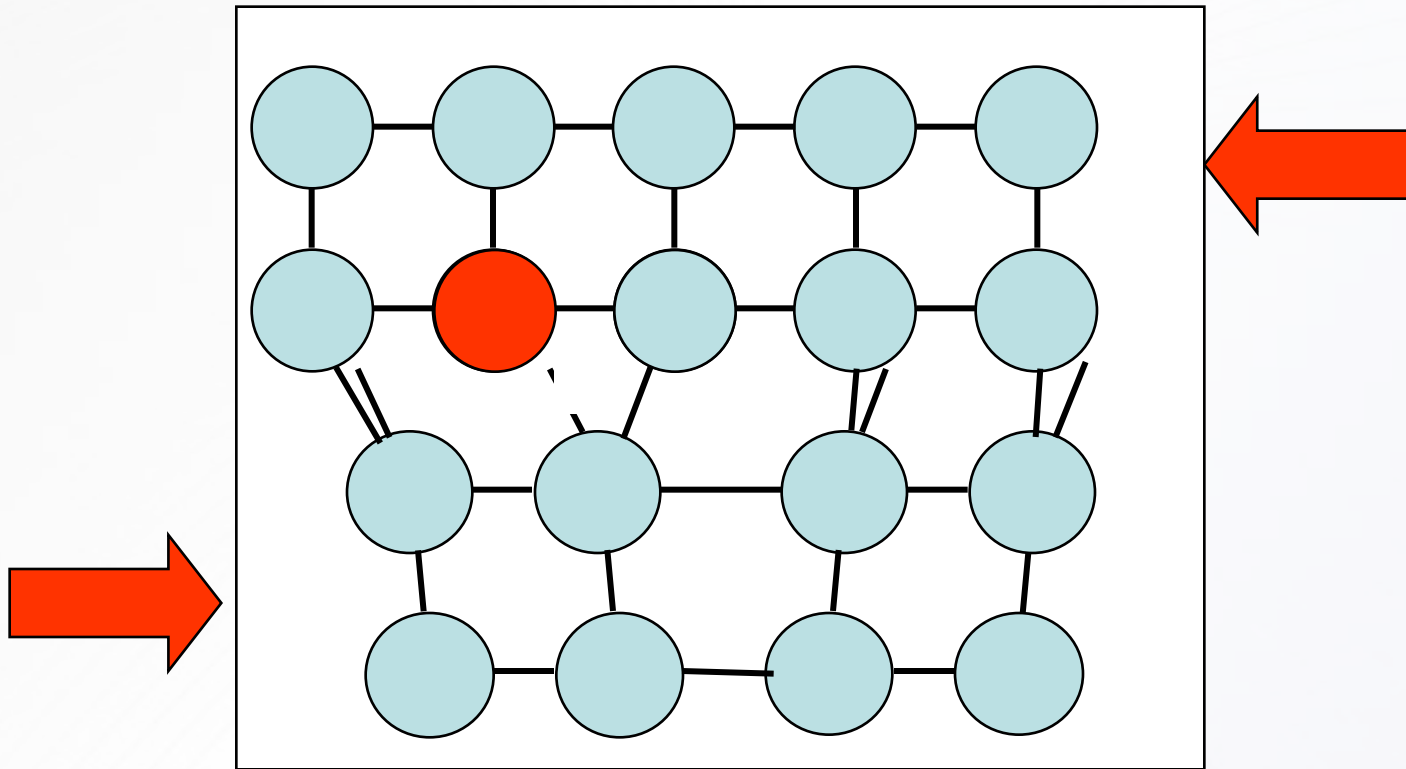
“glide” edge dislocations



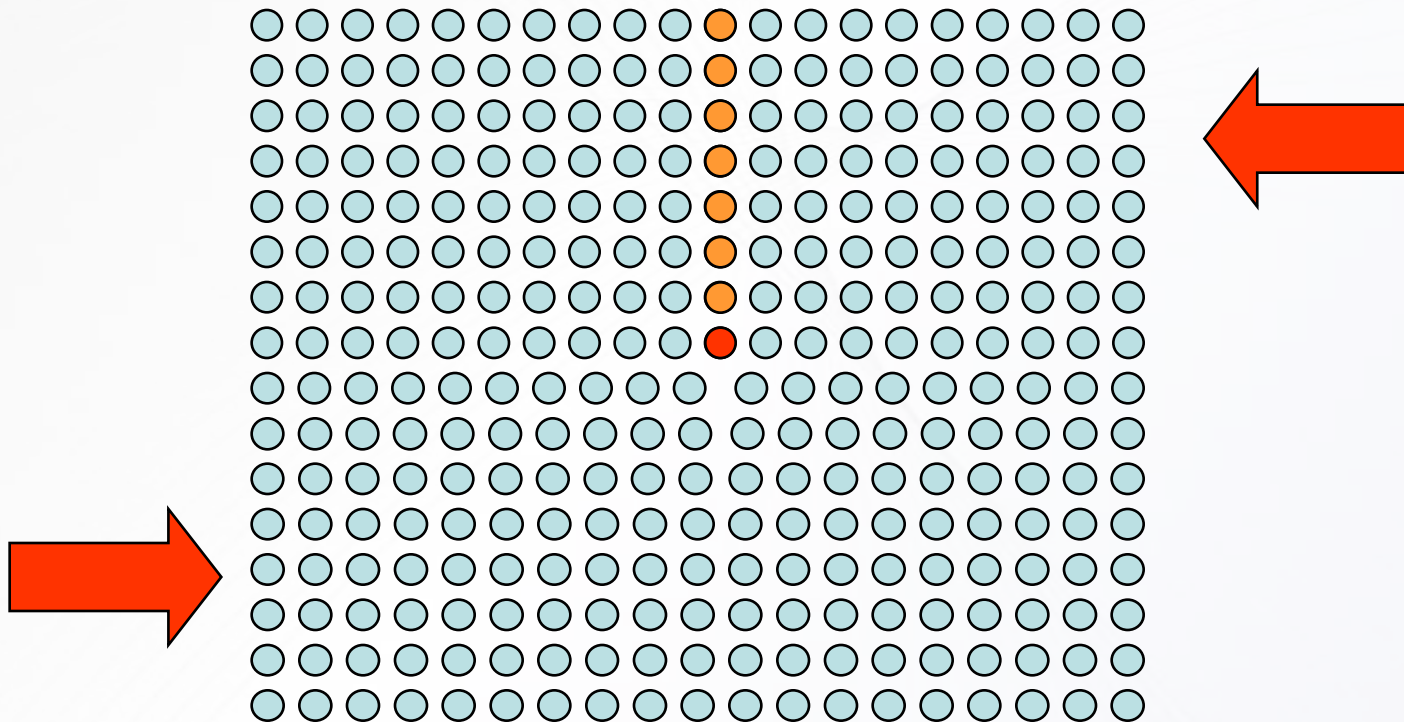
“glide” edge dislocations



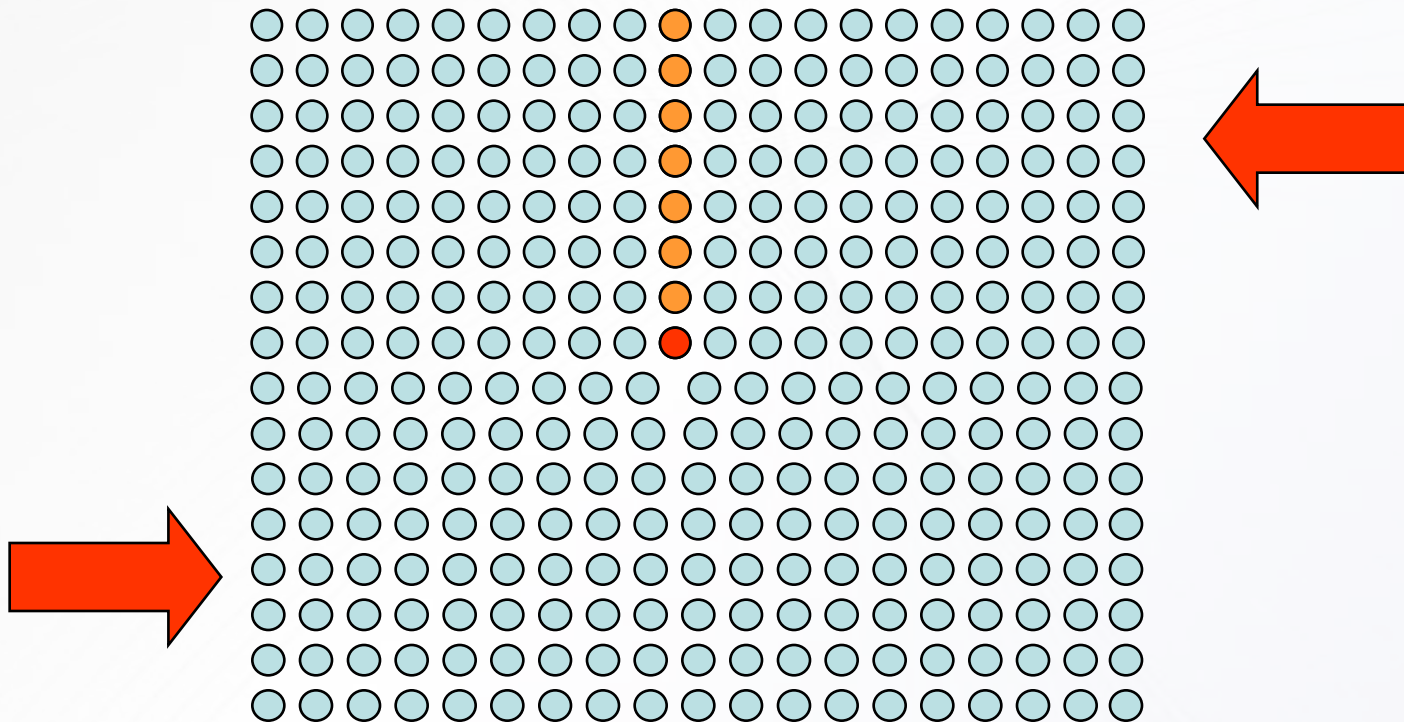
“glide” edge dislocations



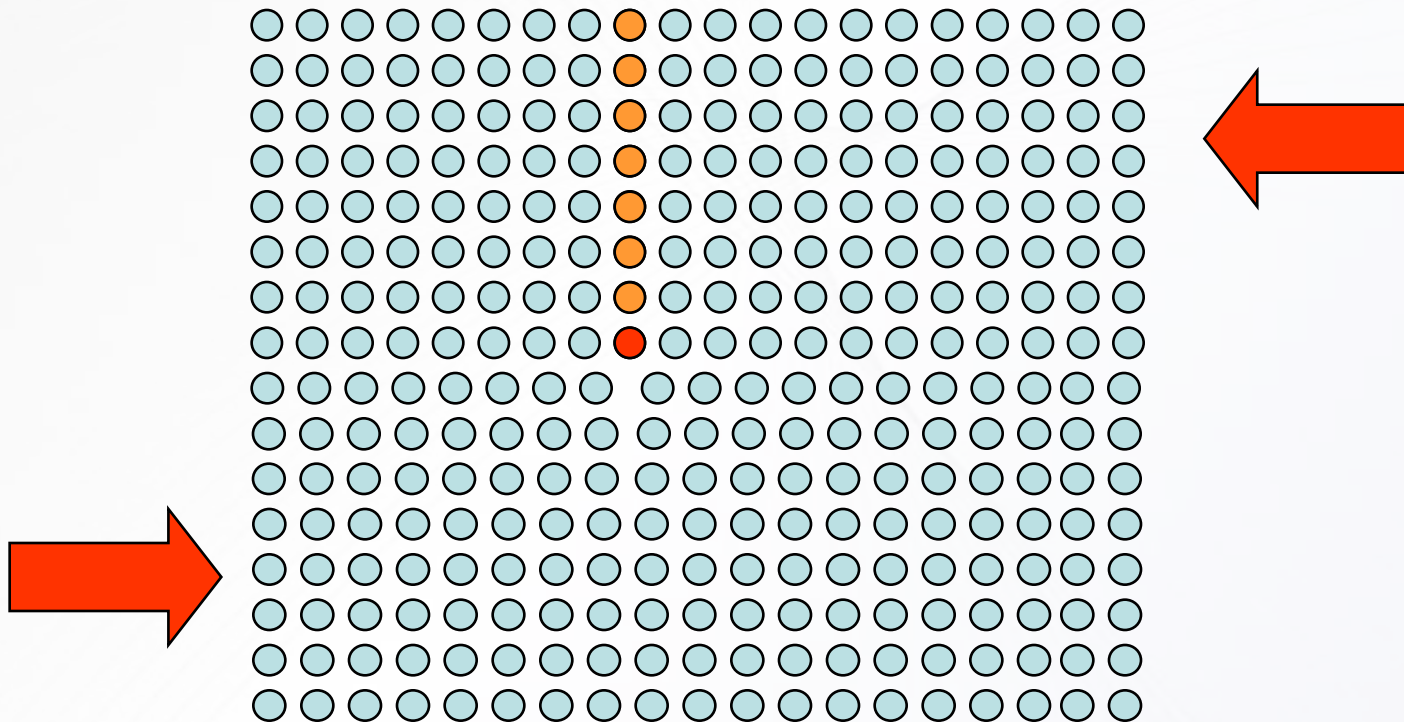
“glide” edge dislocations



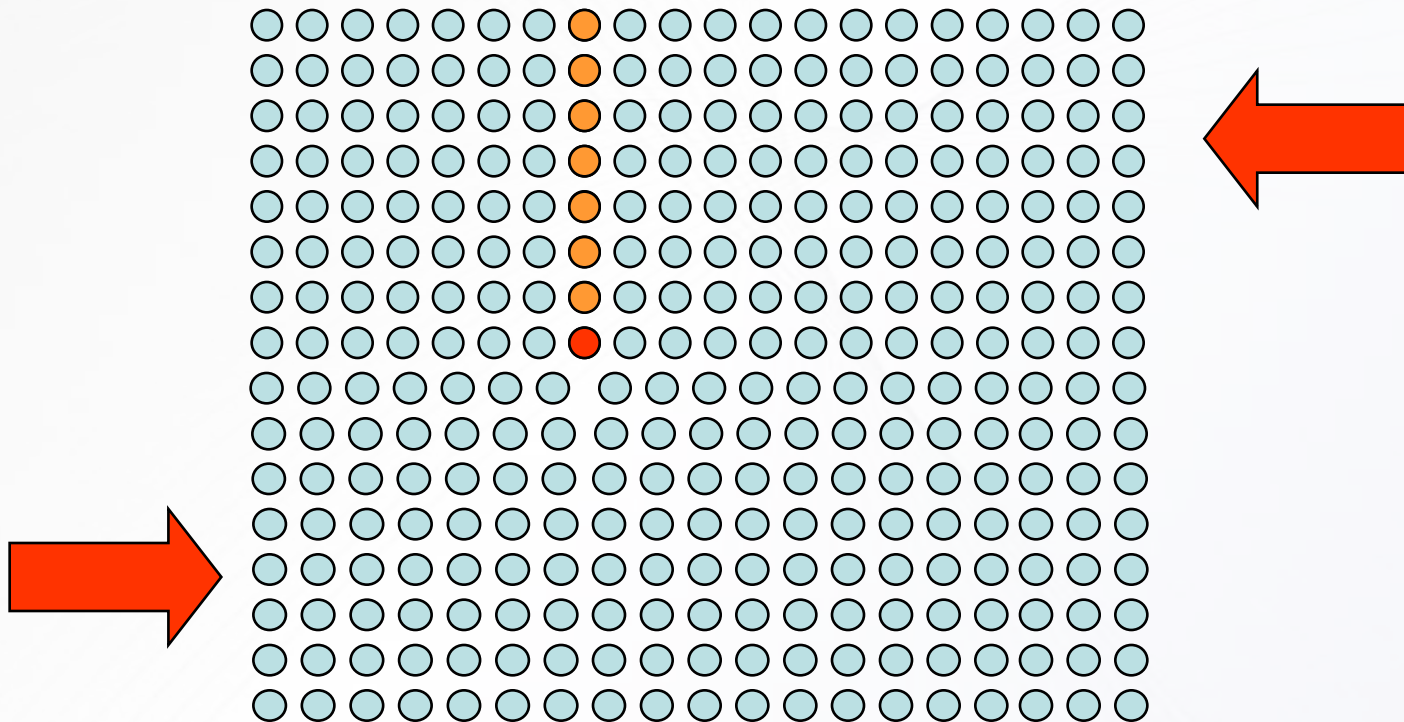
“glide” edge dislocations



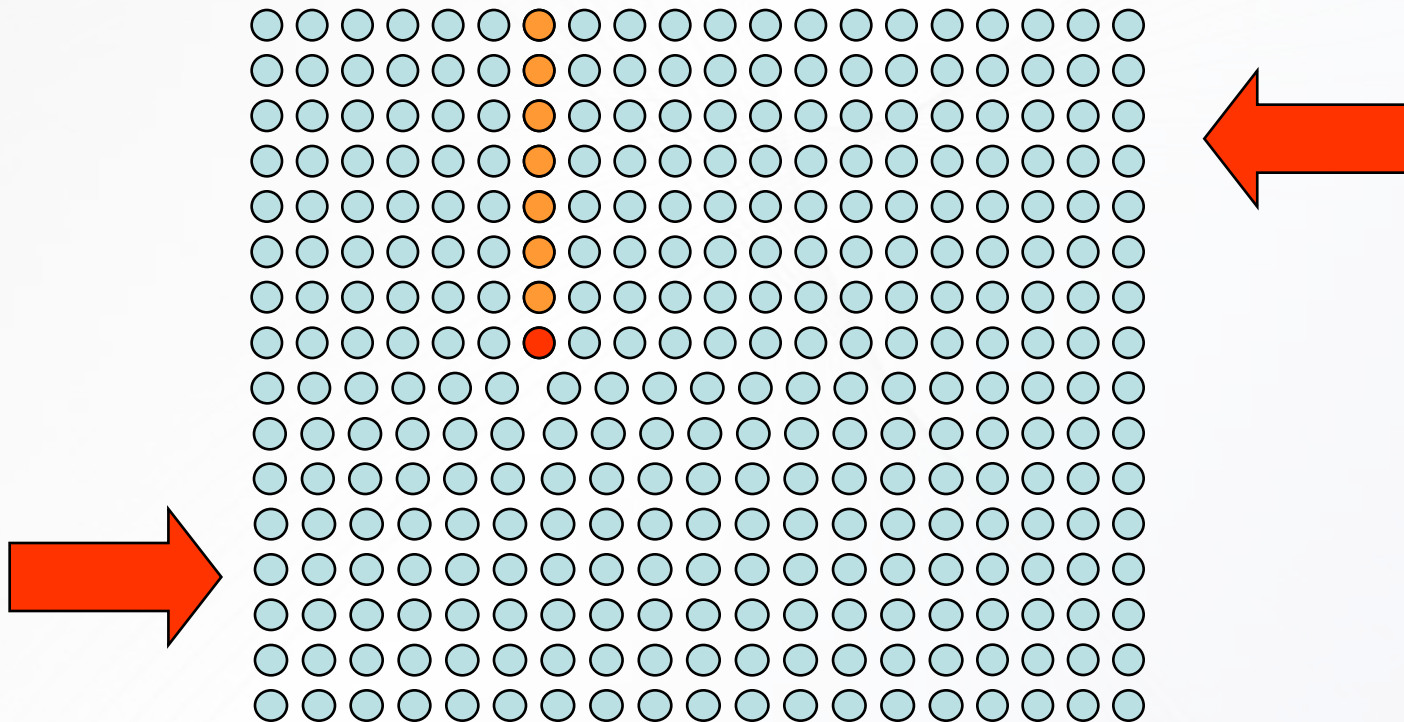
“glide” edge dislocations



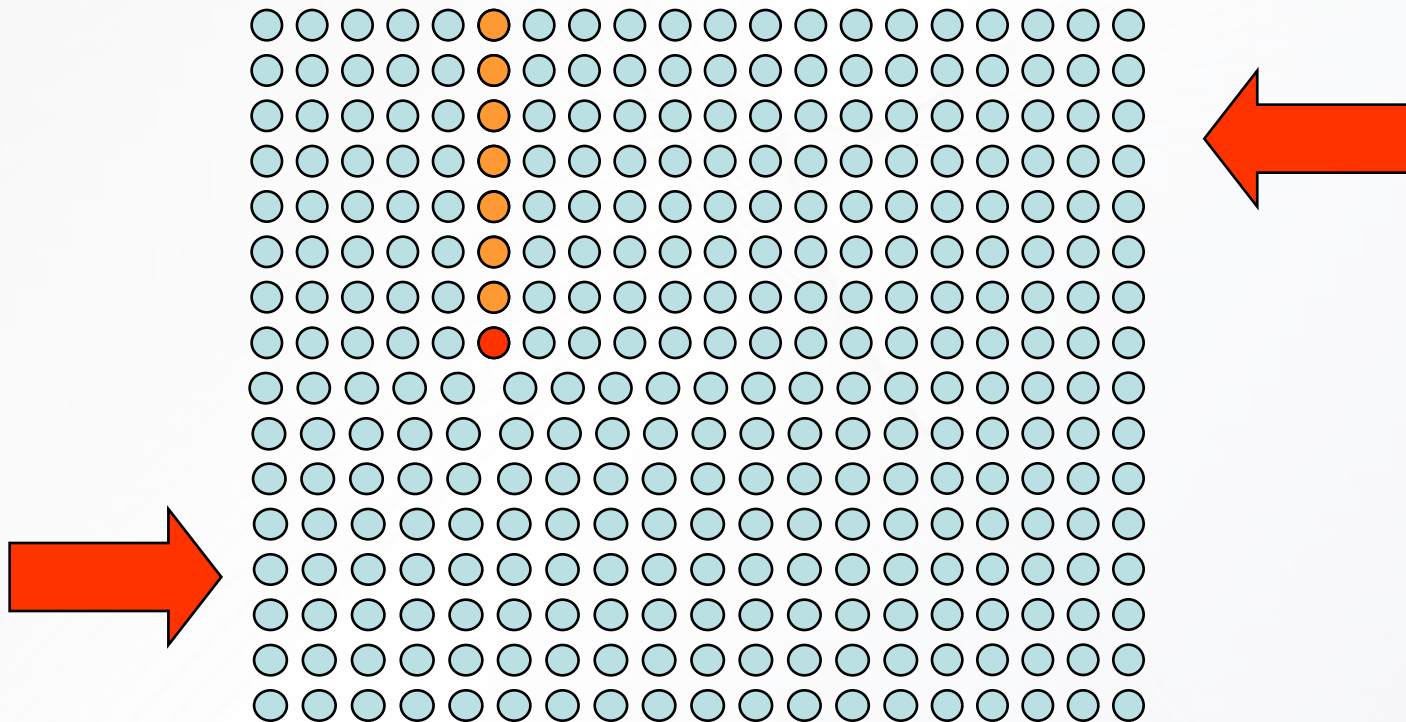
“glide” edge dislocations



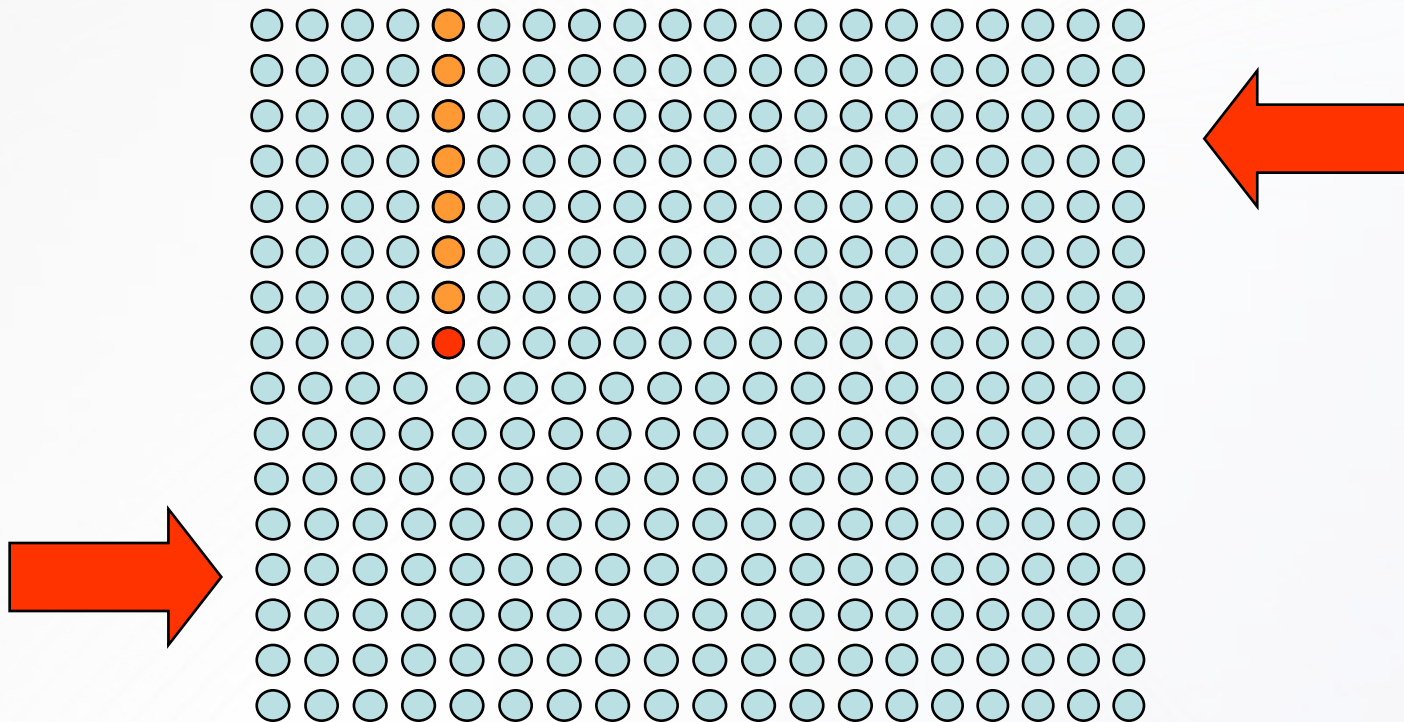
“glide” edge dislocations



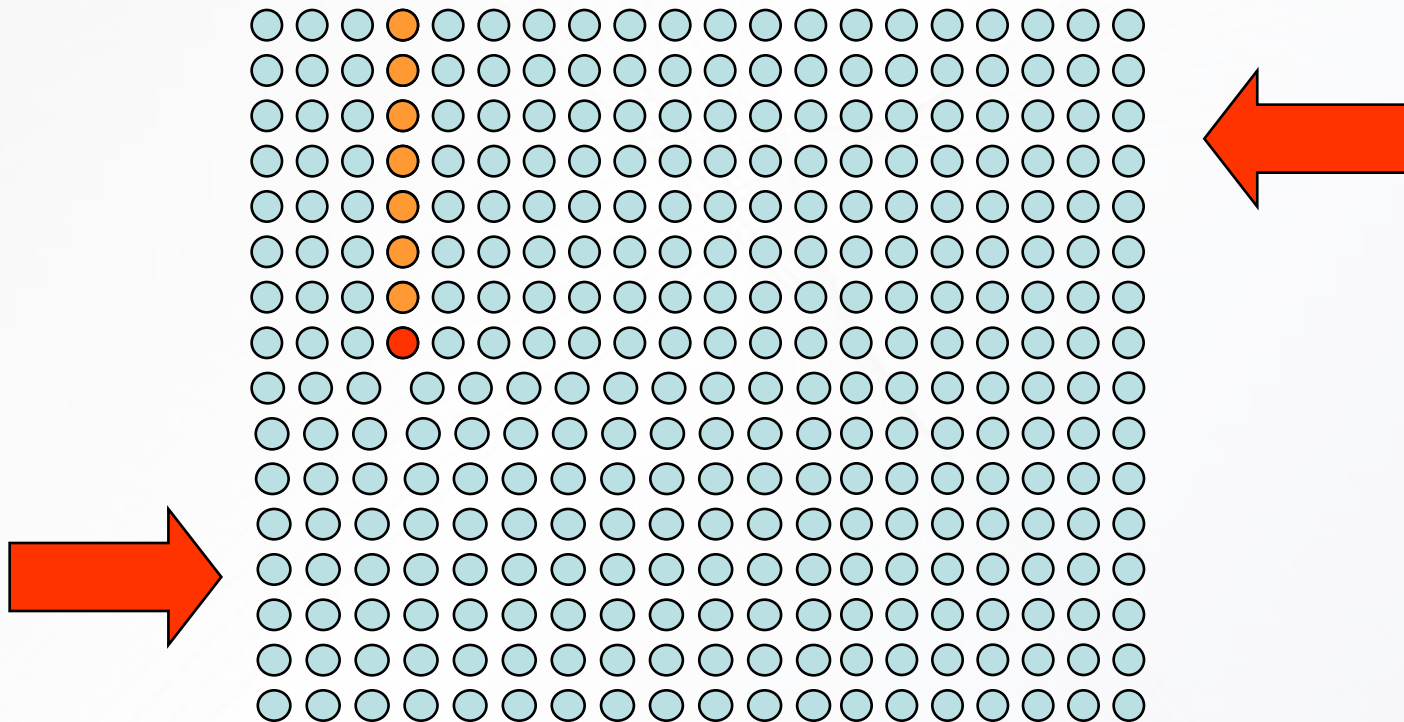
“glide” edge dislocations



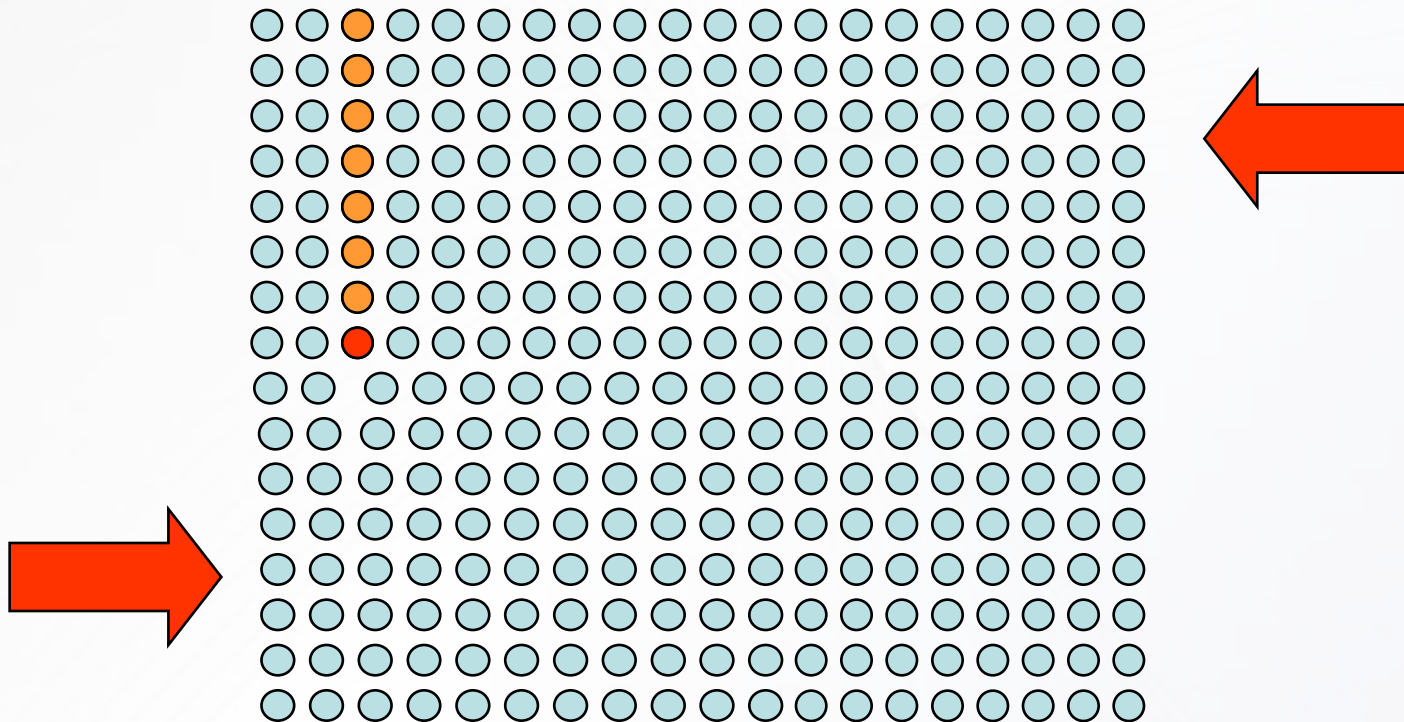
“glide” edge dislocations



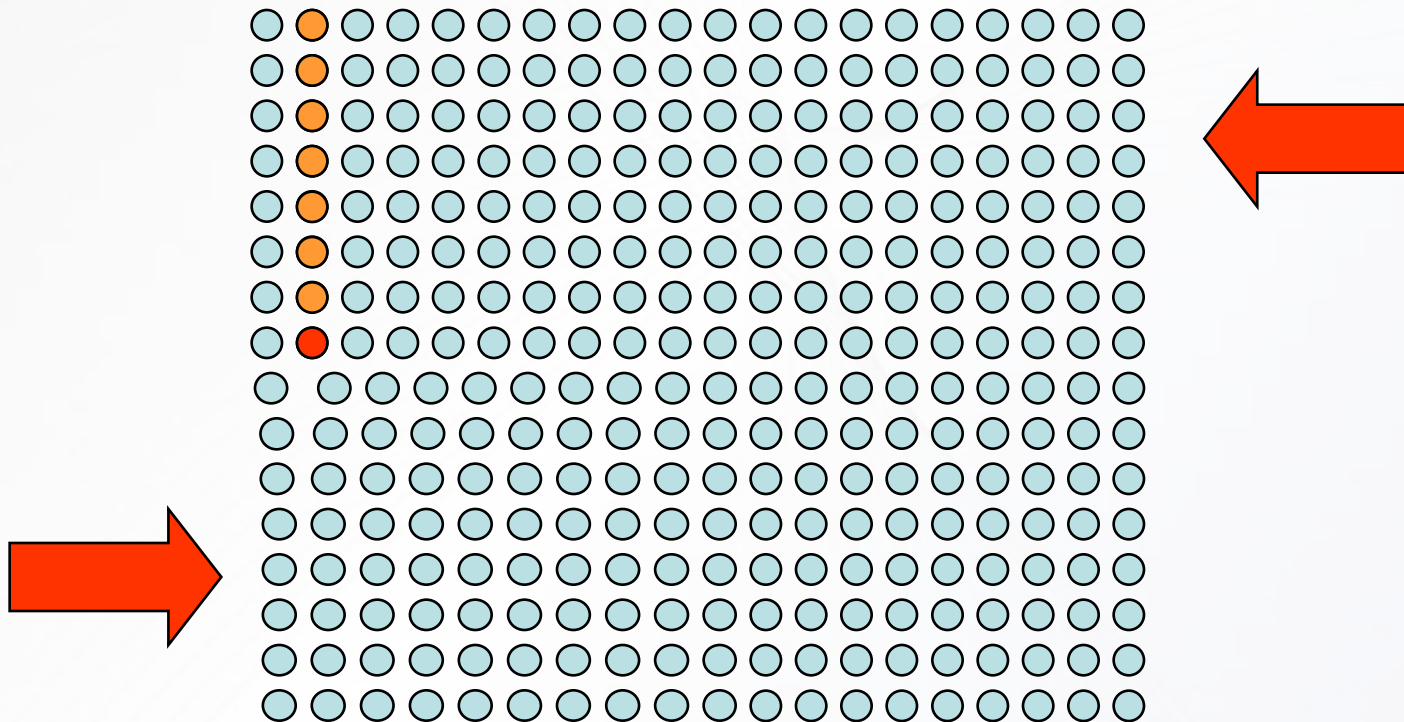
“glide” edge dislocations



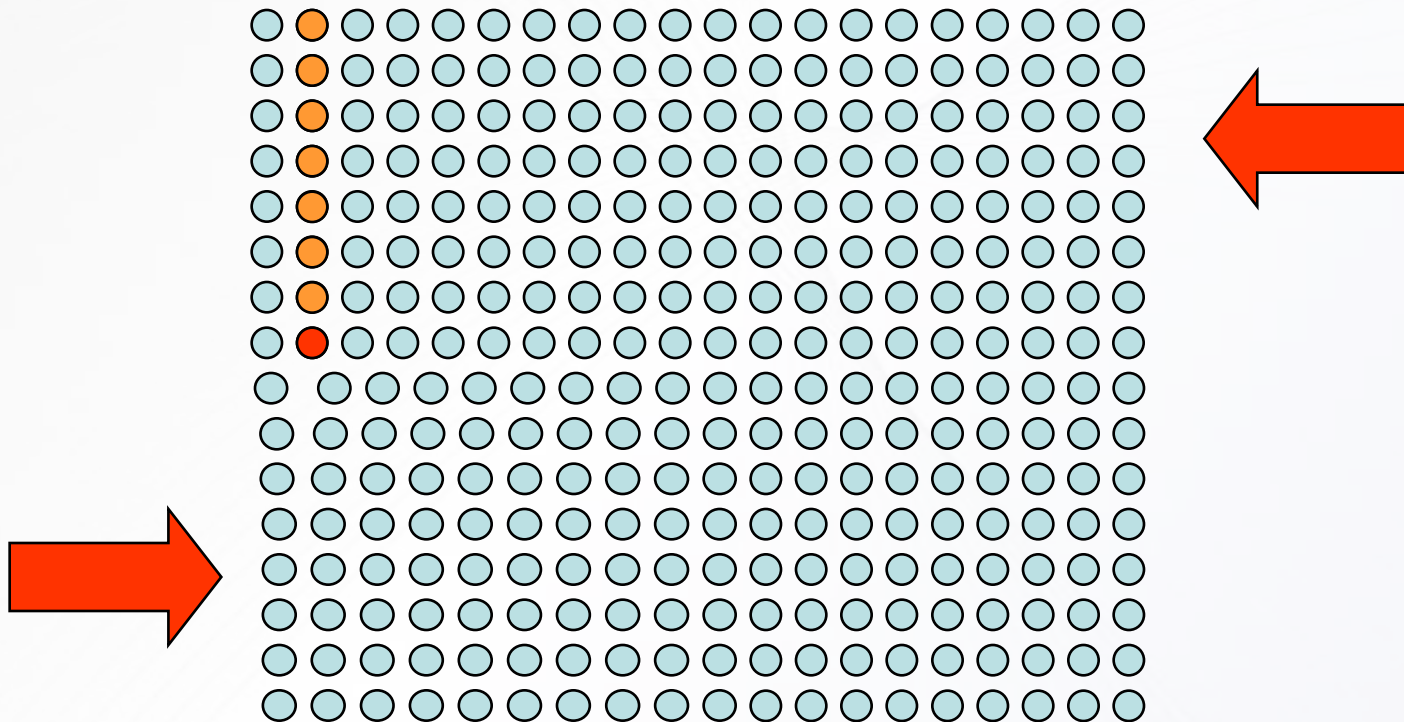
“glide” edge dislocations



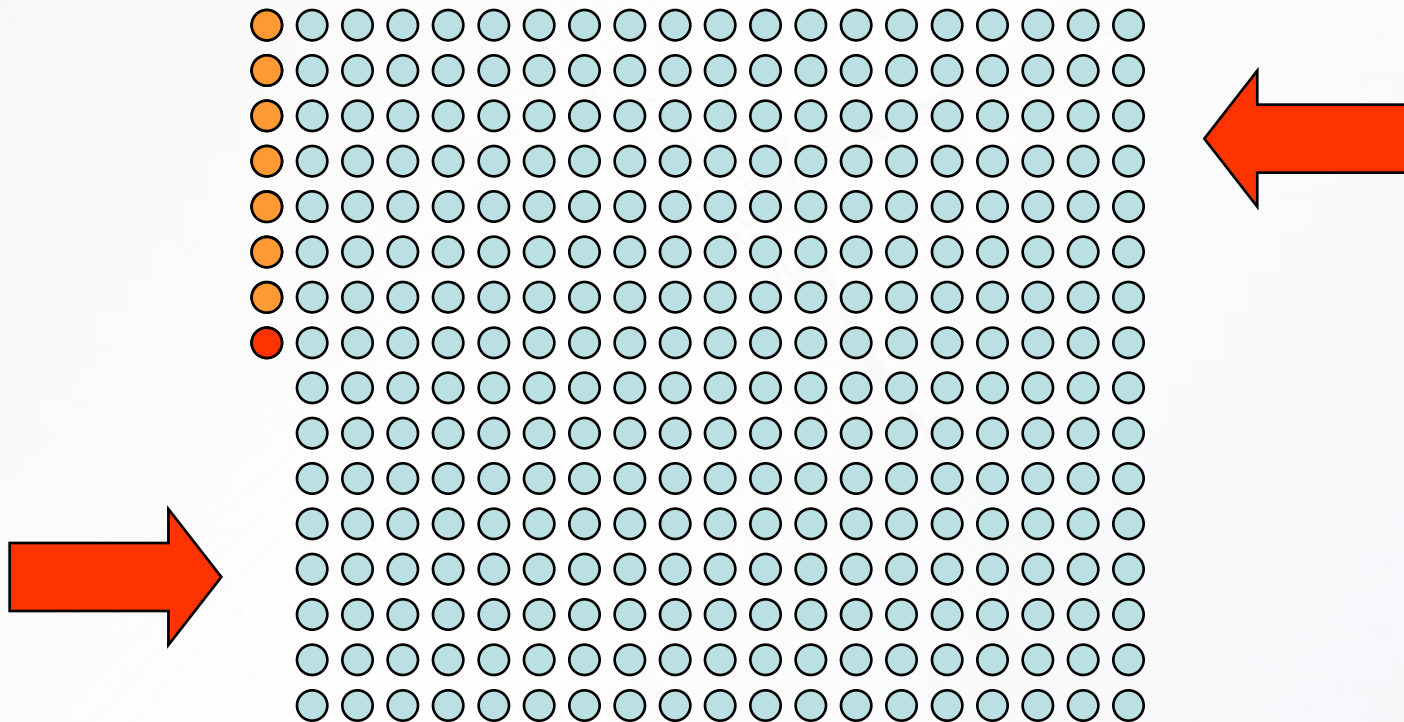
“glide” edge dislocations



“glide” edge dislocations

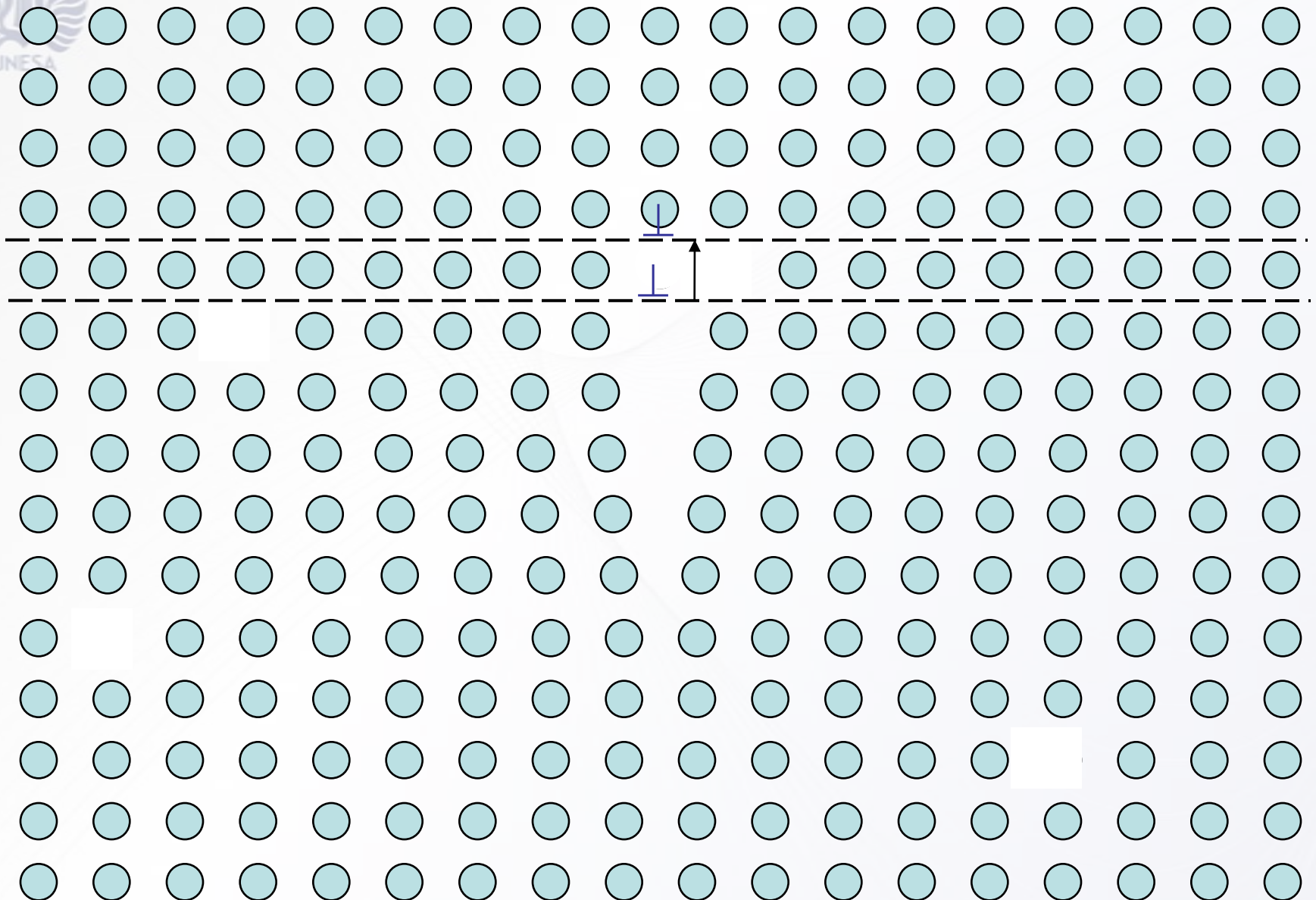


“glide” edge dislocations

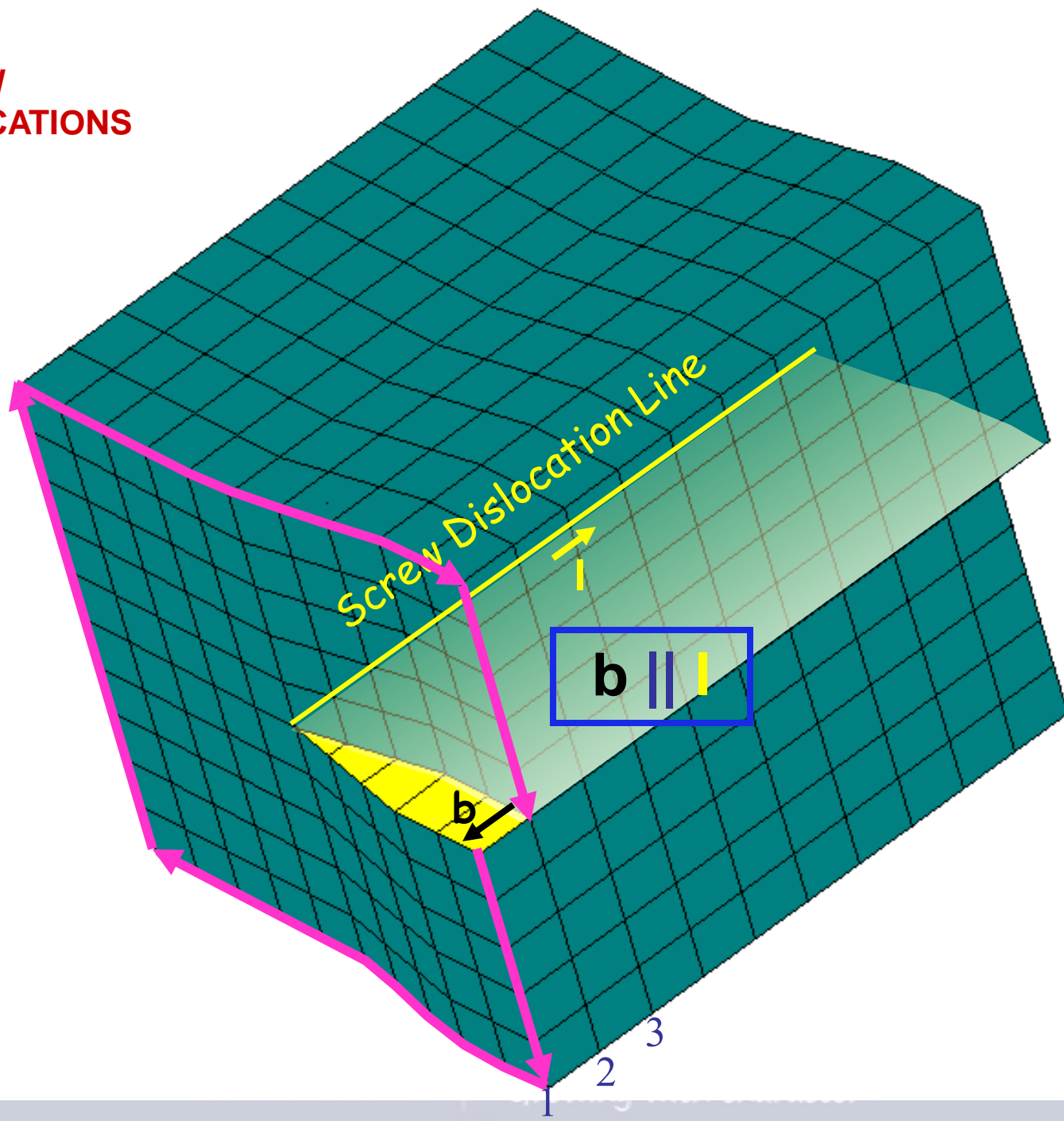


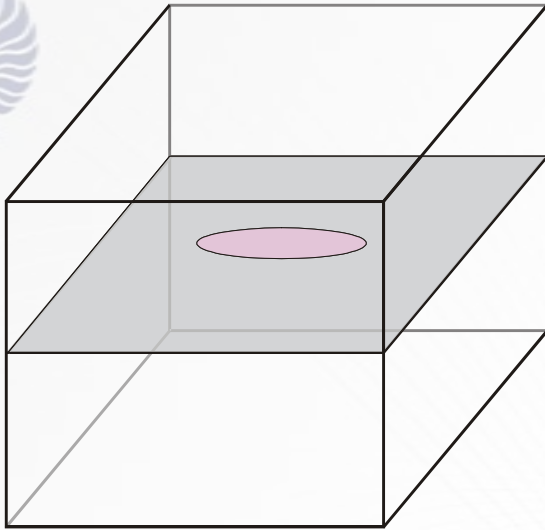


Atomistic mechanism of climb



SCREW DISLOCATIONS





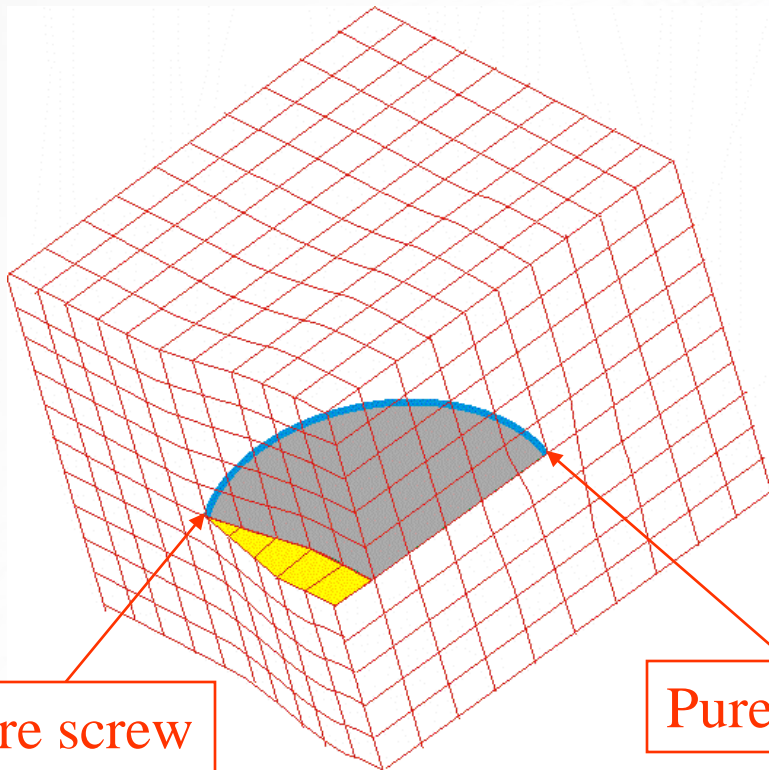
Mixed dislocations



\bar{b}

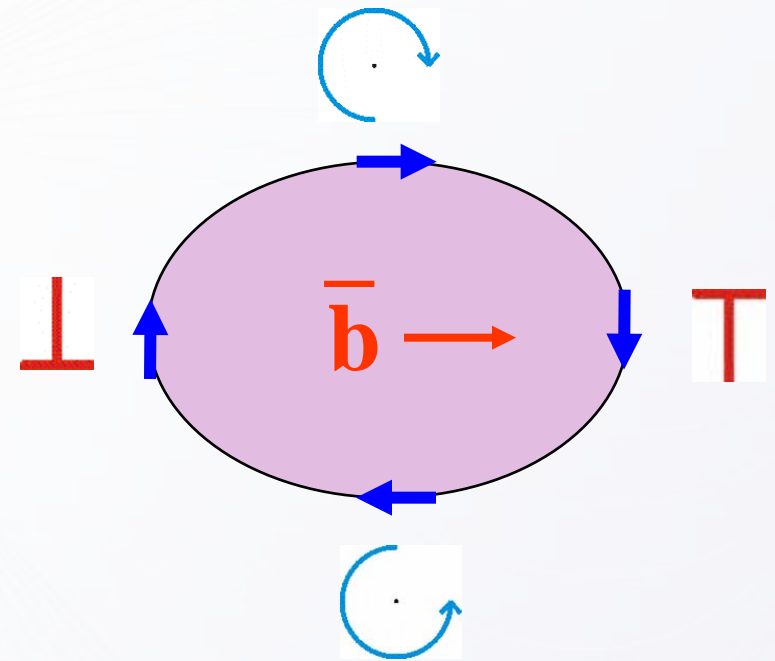


\bar{l}



Pure screw

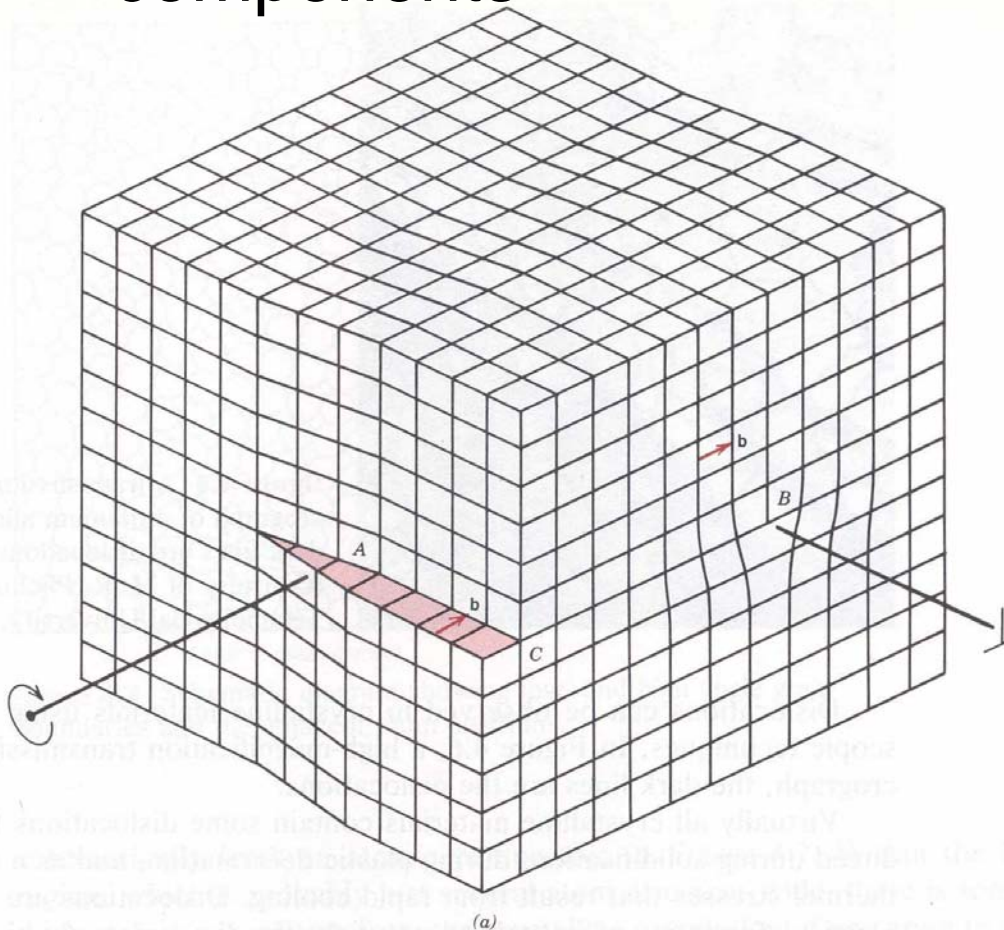
Pure Edge



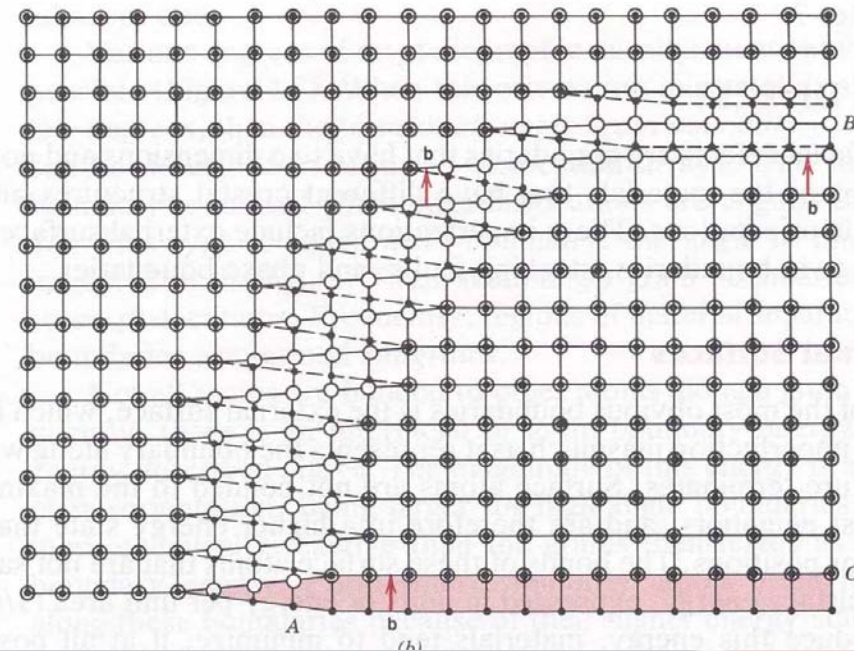
Mixed Dislocations

- Mixed dislocations* have edge & screw components

Orientation of line
w.r.t. fault vector \mathbf{b}
varies along dislocation

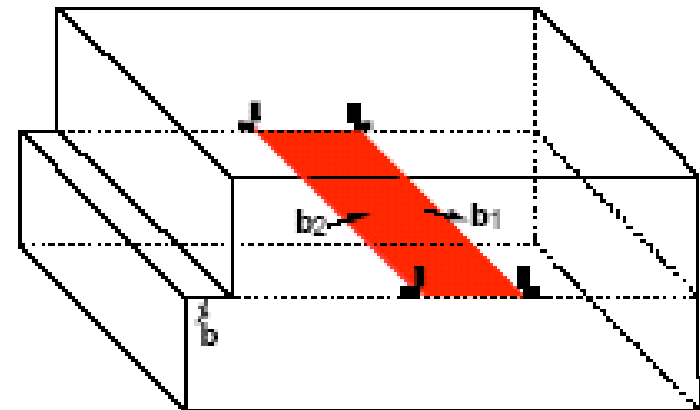
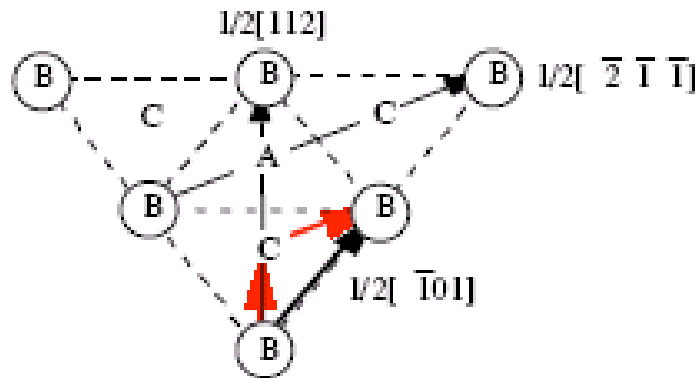


top view





PARTIAL DISLOCATIONS

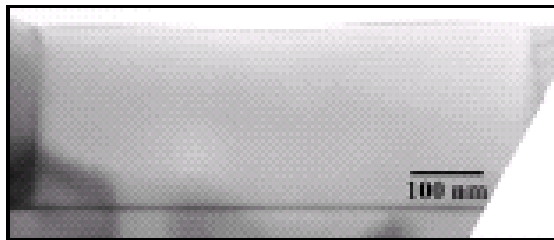
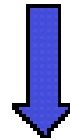


- Sometimes dislocations are split into “partials”
 - Separated by a “stacking fault”
- Common in fcc
 - “Total” dislocation $B \rightarrow B$
 - Splits into “partial” dislocations $B \rightarrow C$ and $C \rightarrow B$
 - Separated by “stacking fault” $\dots ABCABC\dots \rightarrow \dots A|CABCA\dots$

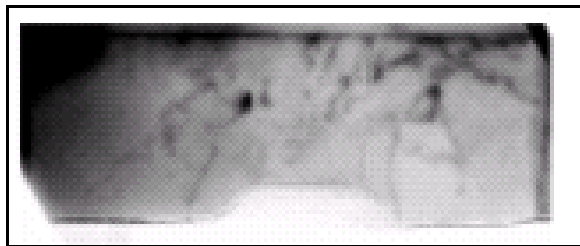
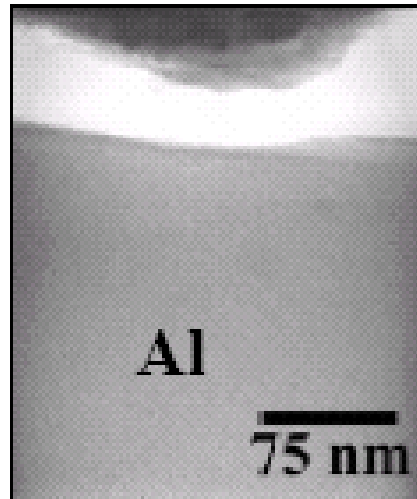
Dislocations Created on Indenting Al

- Minor, Stach and Morris, *Appl. Phys. Lett.*, 79, 1625-27 (2001)

Diamond Nanoindenter



Before

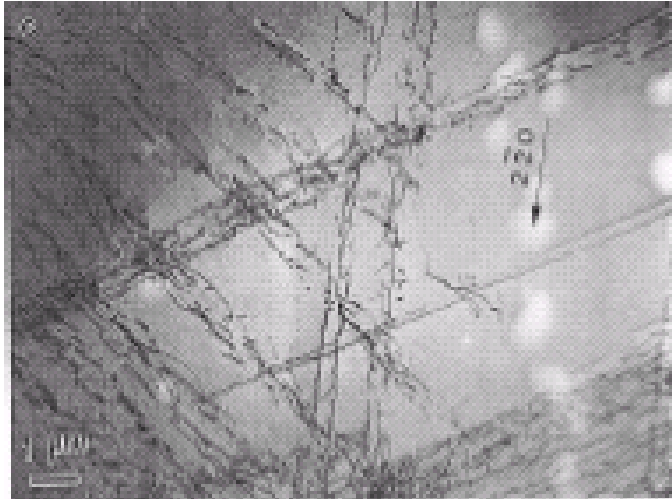


After



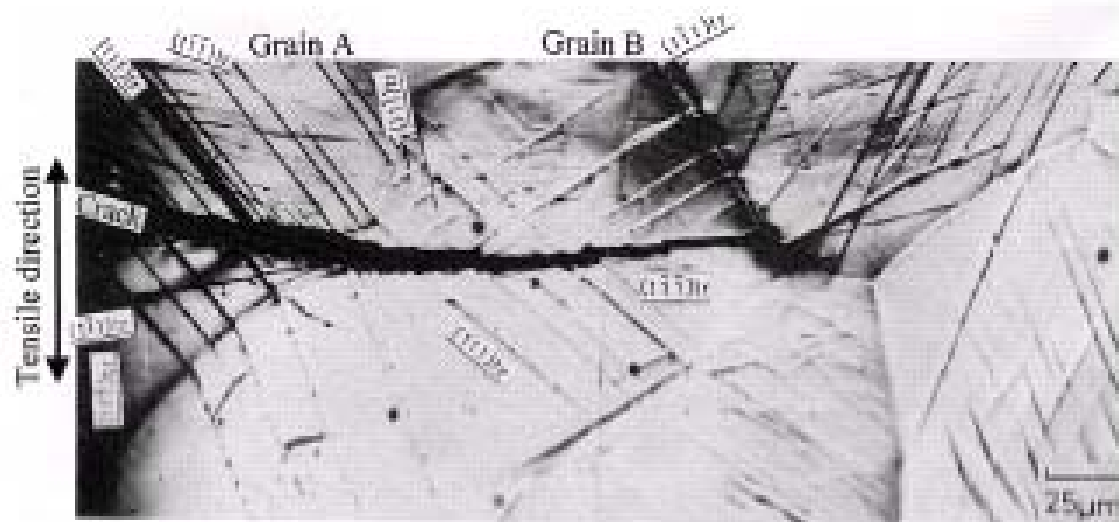
Dislocations in TiC

- Chien, Ning and Heuer, *Acta Mater.*, 44, 2265 (1996)



Slip Lines Across Grains in Stainless Steel

- Tomota, et al., *Acta Mater.*, 46, 3099-3108 (1998)





Defects

Dimensionality

Examples

Point

0

Vacancy

Line

1

Dislocation

Surface

2

Free surface,
Grain boundary

Volume

3

Voids, Inclusions,
Precipitates

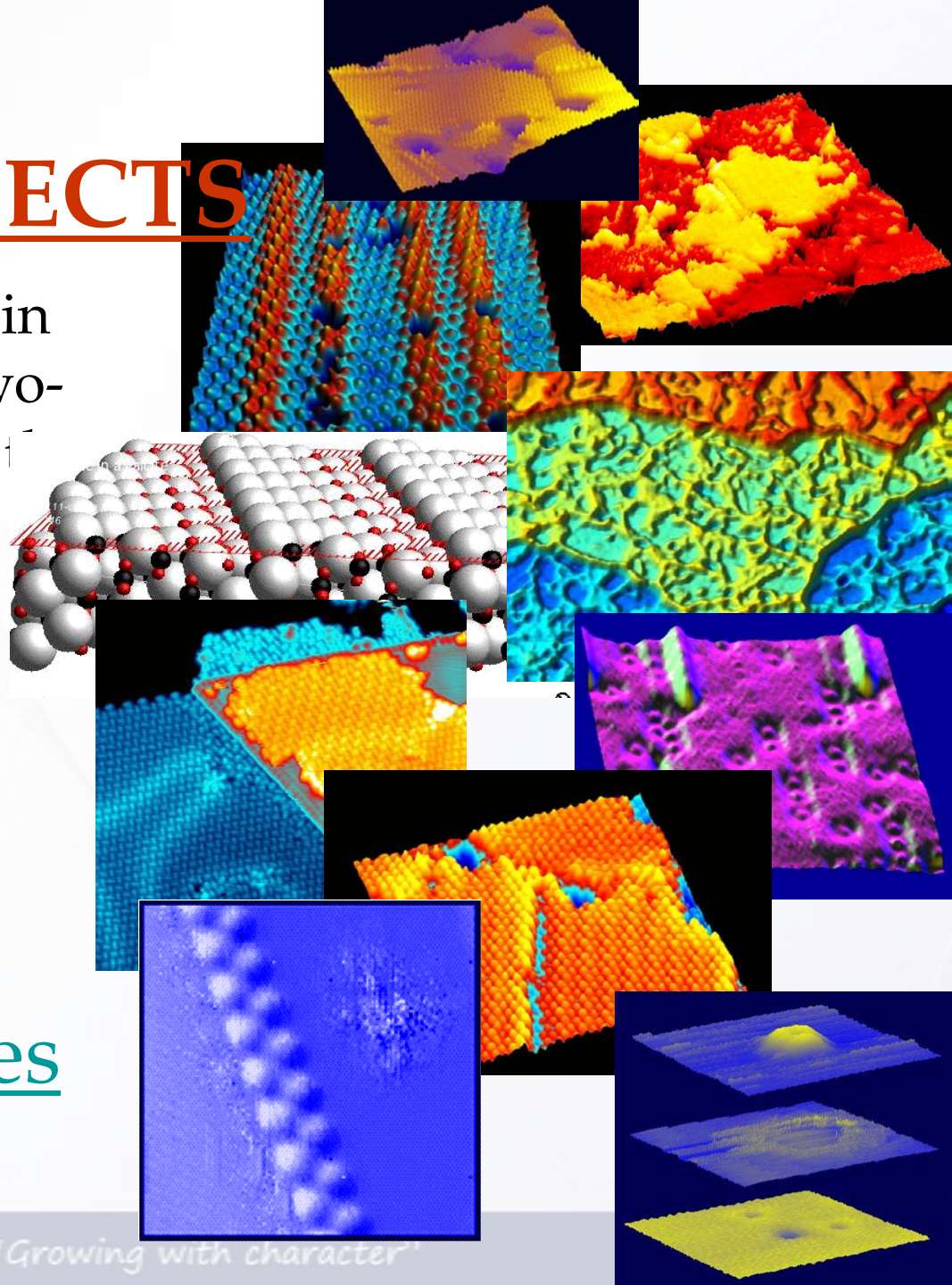


SURFACE DEFECTS

Imperfections, such as grain boundaries, that form a two-dimensional plane within a crystal.

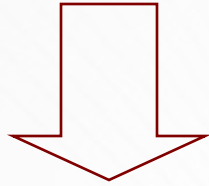
classifications

- free surface
- twin boundary
- stacking faults
- grain boundaries





VOLUME DEFECTS



Three-dimensional defects in solids

Volume defects play an important role in corrosion mechanisms

Always involve a second phase

- Porosity (solid – vapor)
- Inclusions (solid – solid)
- Precipitates (solid – solid)
- Cracks (solid – vapor)

- voids
- inclusions
- precipitates

ENTER

External

Internal

Free surface

Grain boundary

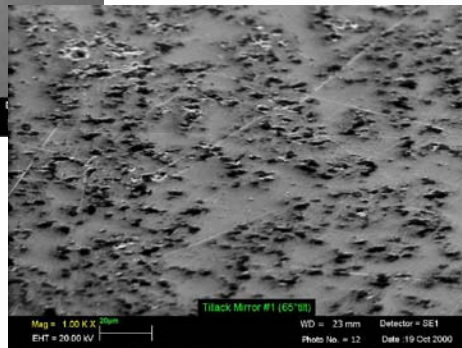
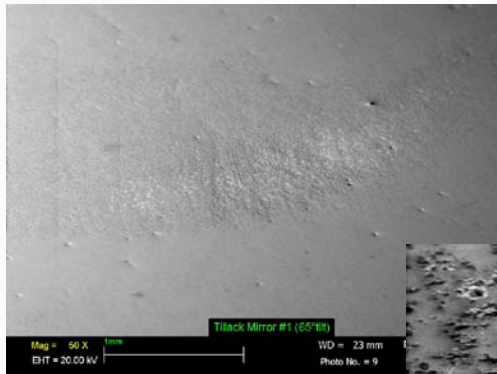
Stacking fault

*Same
phase*

Twin boundary

Interphase boundary

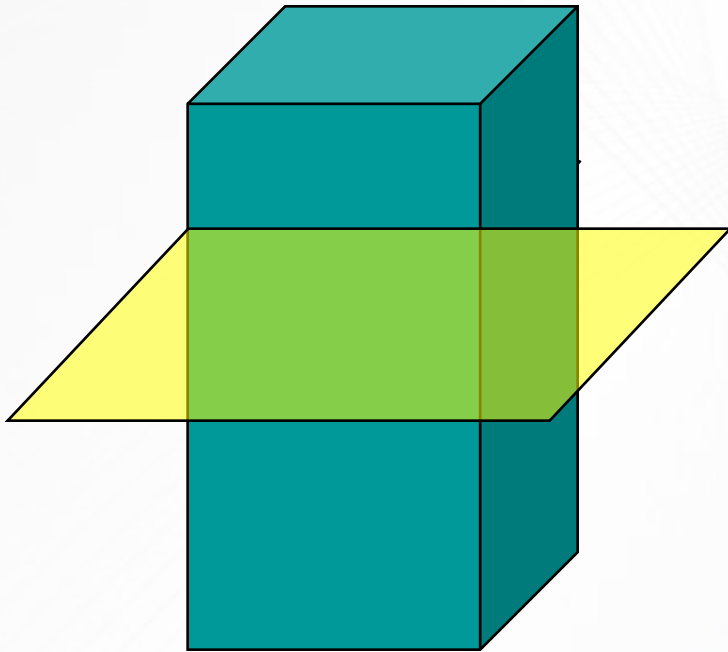
*Different
phases*



bome

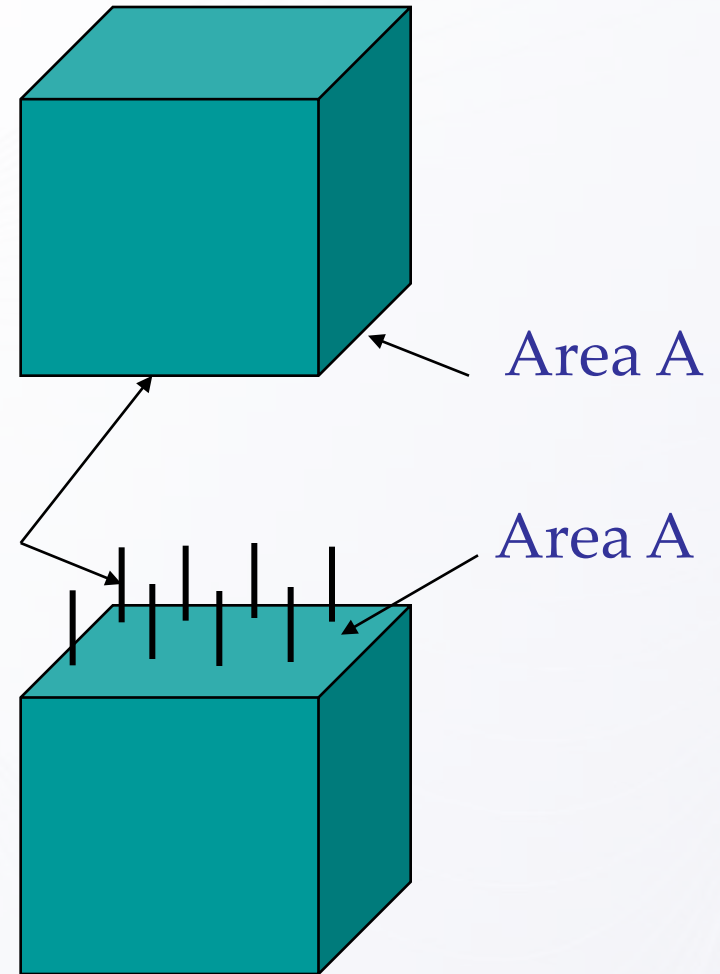
free surface

Surface grooving where grain boundaries intersect free surfaces leads to surface roughness, possibly break-up of thin films



If bond are broken over an area A
then two free surfaces of a total
area $2A$ is created

Broken
bonds



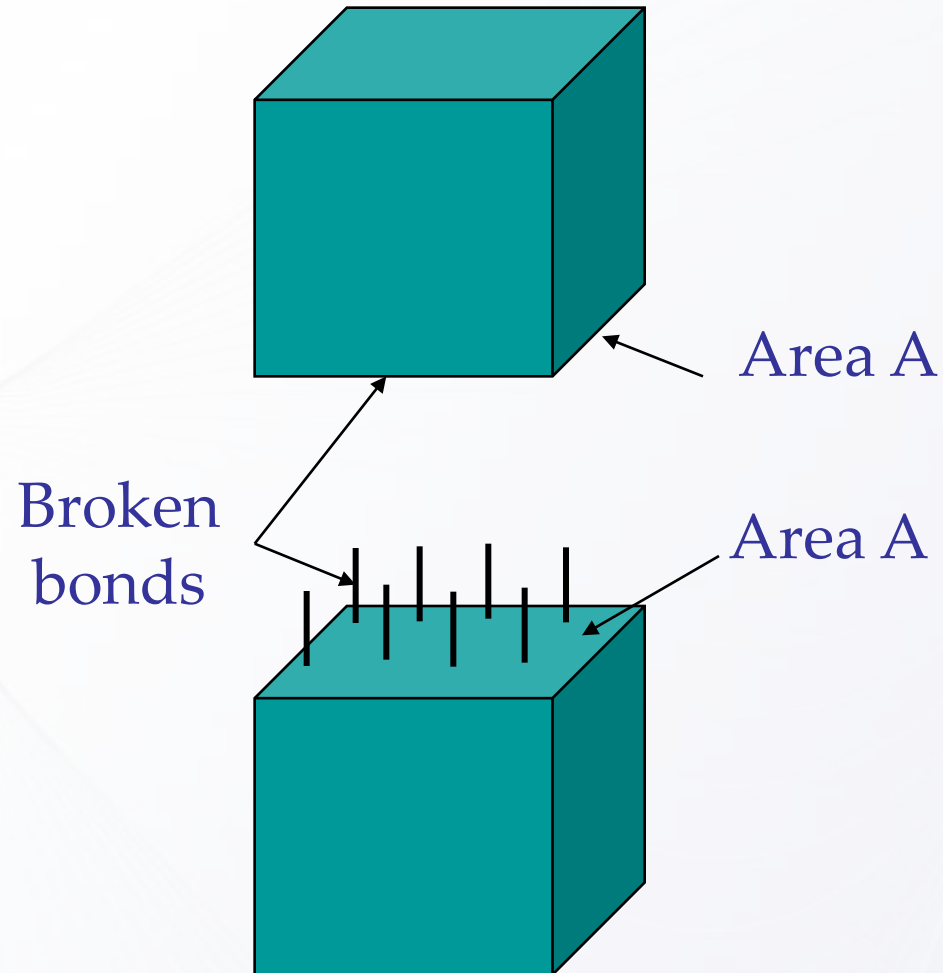
n_A = no. of surface atoms per unit area

n_B = no. of broken bonds per surface atom

ε = bond energy per atom

$$\gamma = \frac{1}{2} n_A n_B \varepsilon$$

Surface energy per unit area

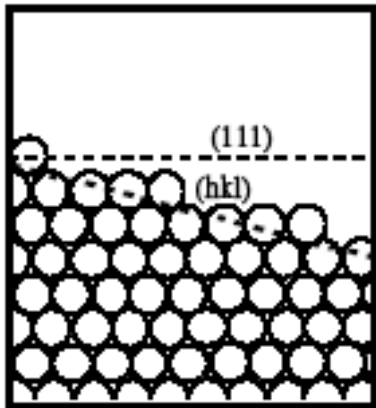


If bond are broken over an area A then two free surfaces of a total area 2A is created

Surface energy is anisotropic

Surface energy depends on the orientation, i.e., the Miller indices of the free surface

n_A, n_B are different for different surfaces

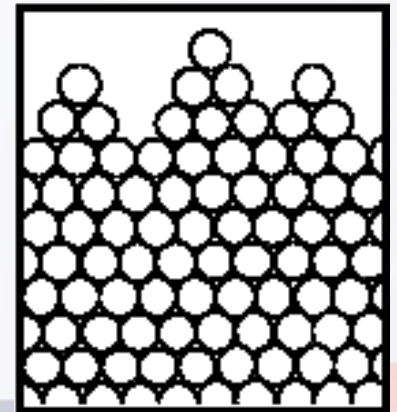


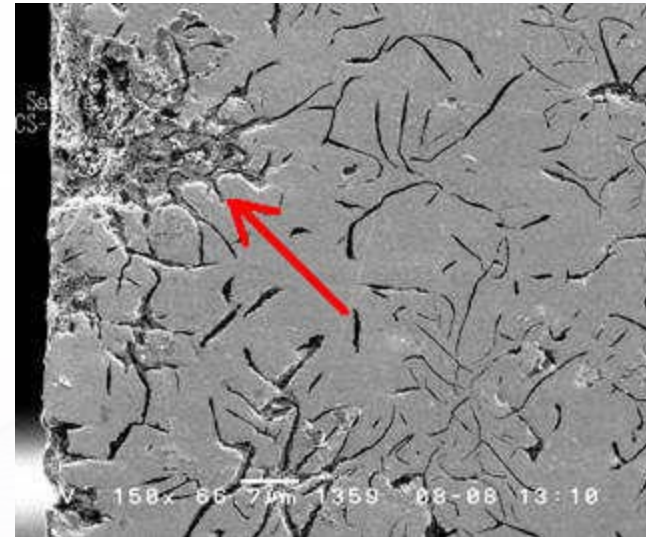
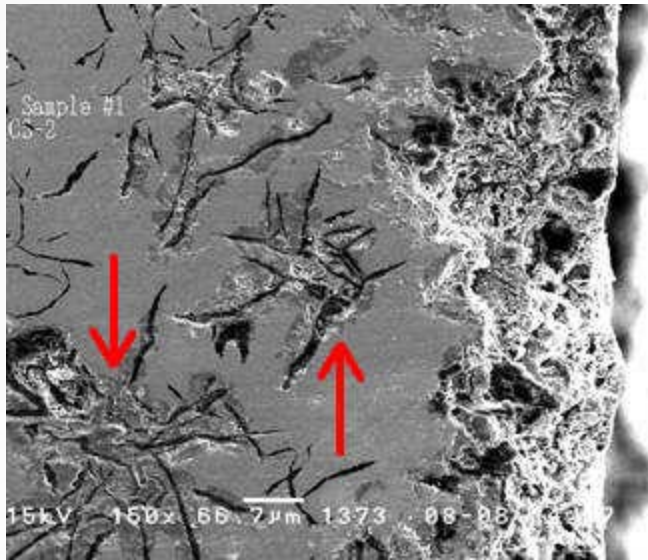
Free Surfaces of Metals

Surface tension (σ) lowest for low-index planes

Diffuse Interface

At high T, metal surfaces tend to be rough, diffuse





free surface

bome





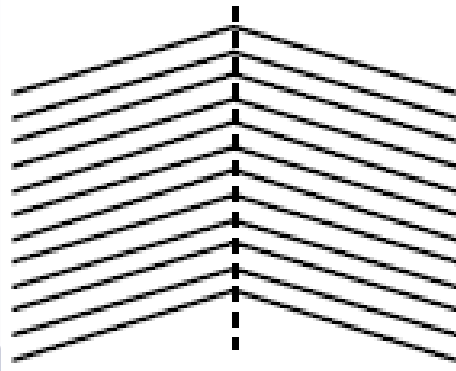
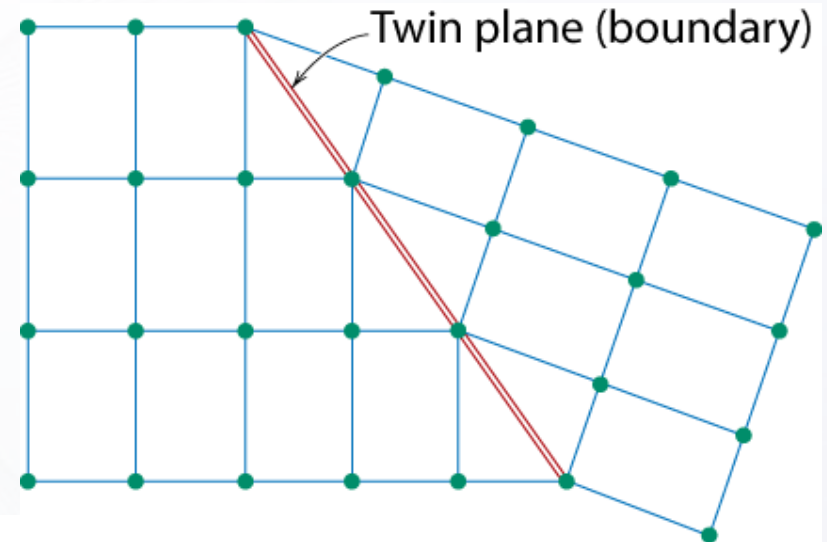
twin boundary (plane)

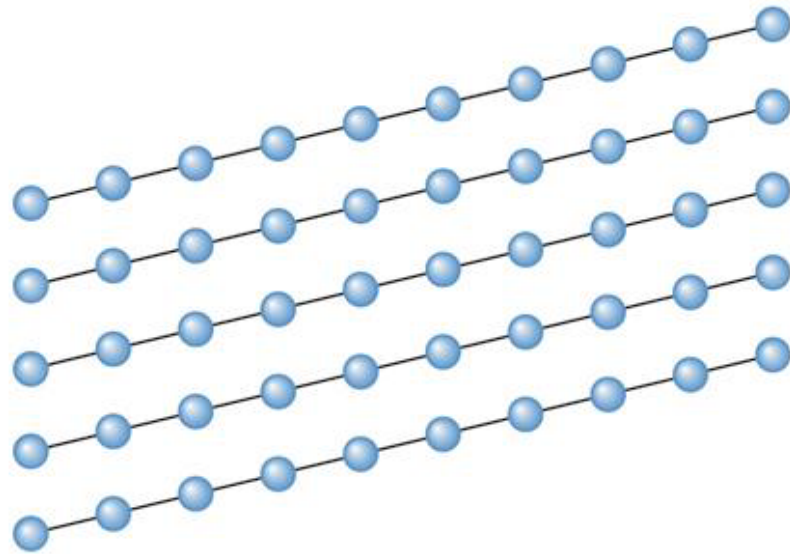
Essentially a reflection of atom positions across the **twin plane**

Twining is very common in minerals (result of phase transition during cooling)

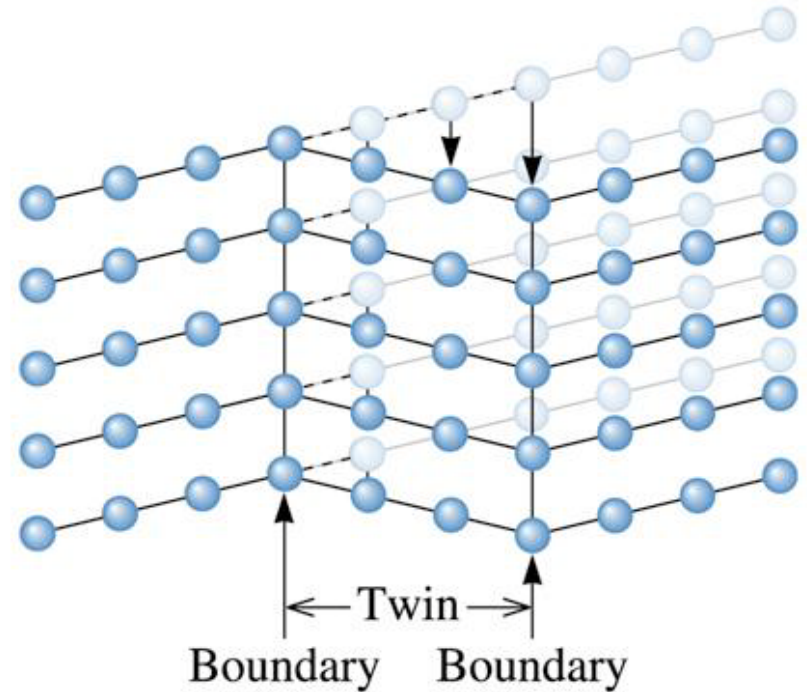
Twinning is an important deformation mechanism

Sn, Mg, high-N
austenitic (FCC) steel,
Cu at low T



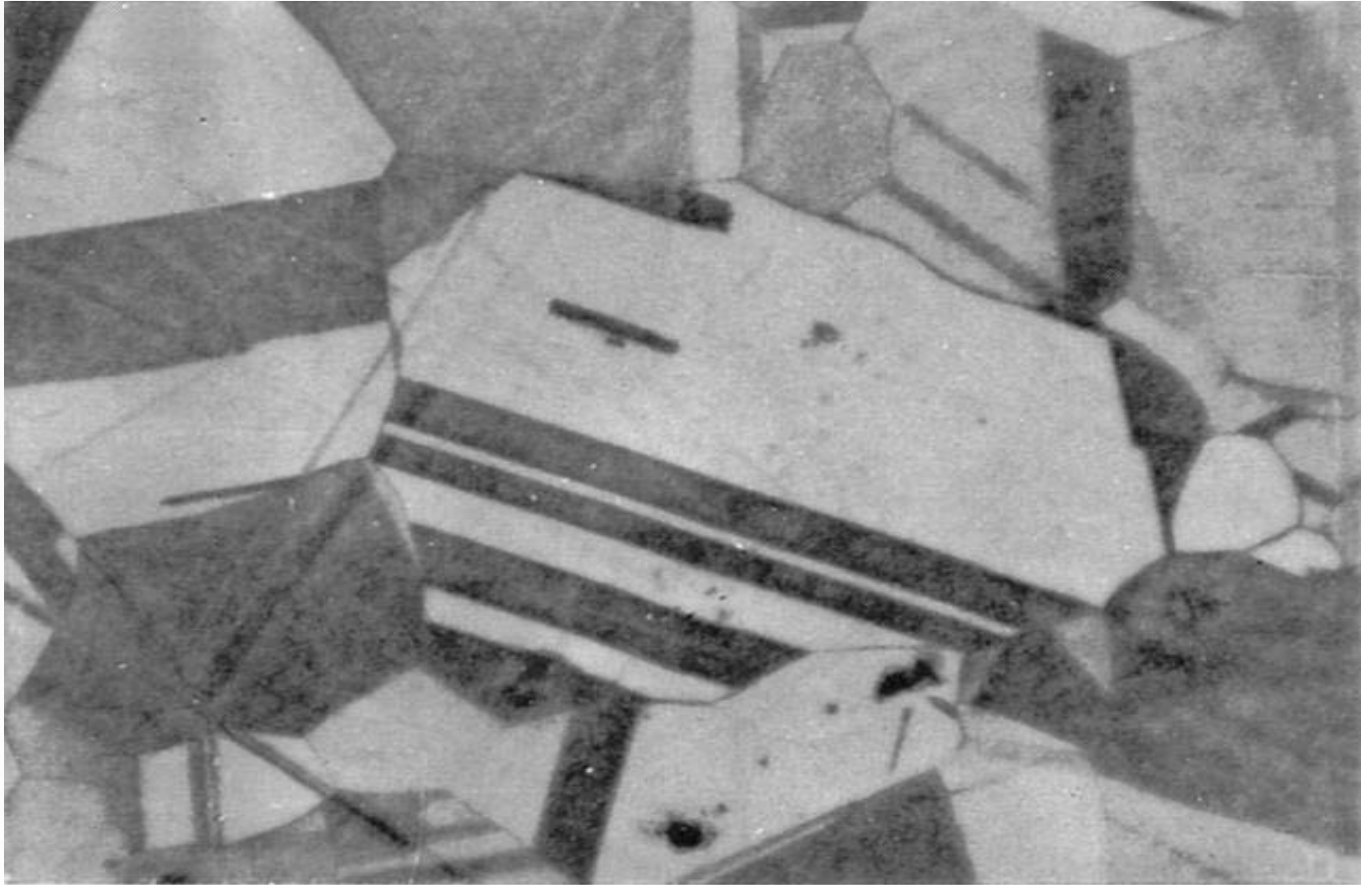


(a)



(b)

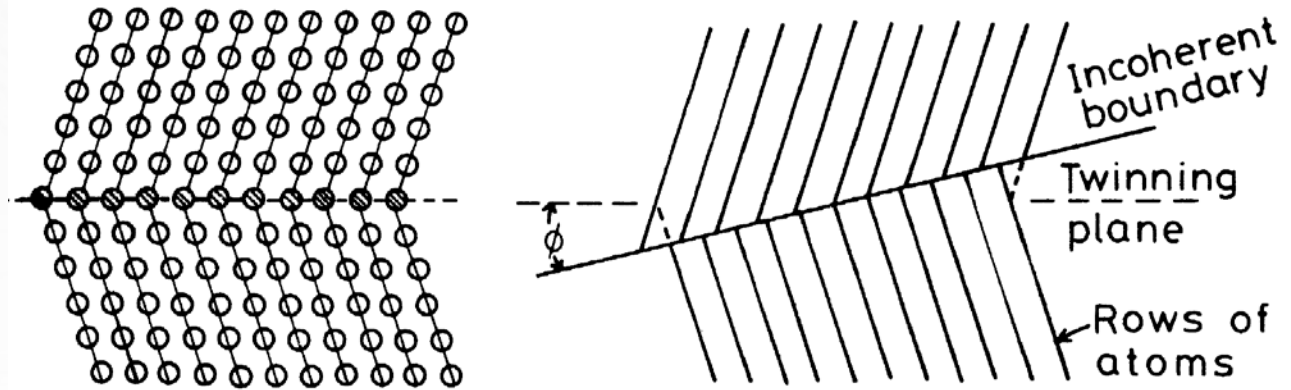
Formation of twin (b) may be caused by application of stress to the perfect crystal (a)



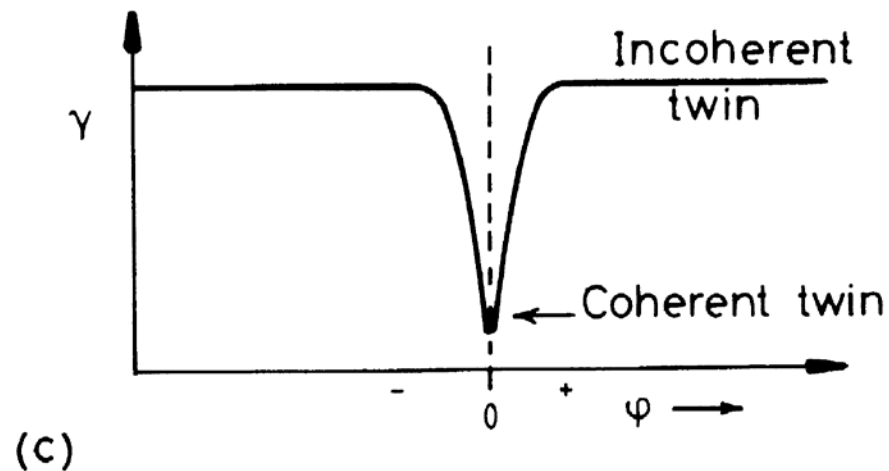
(c)

Figure (c) : A micrograph of twins within a grain of brass (x250)

Twin: coherent vs. incoherent



(b)



(c)

(Porter & Easterling -
fig.3.12/p123)

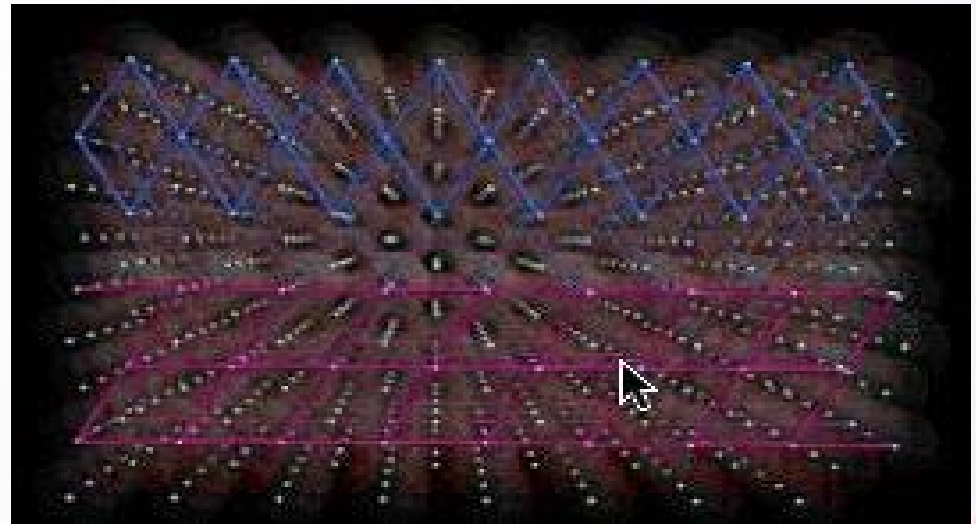


stacking faults

- Stack close-packed planes in wrong sequences
Create extra or missing plane inside the crystal

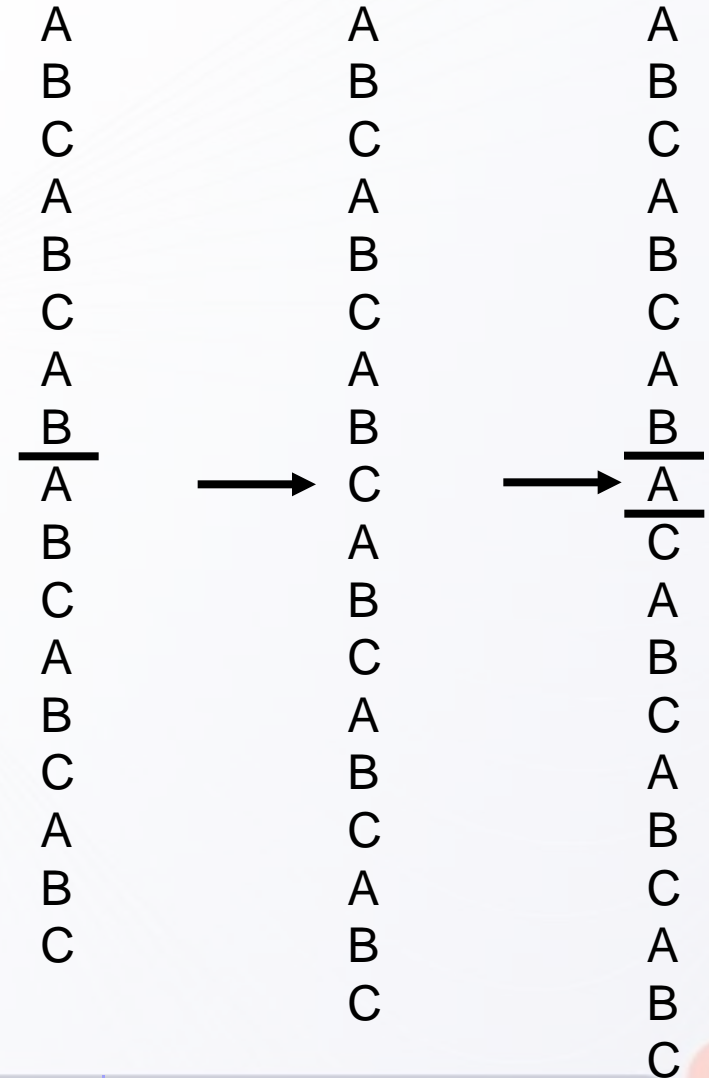
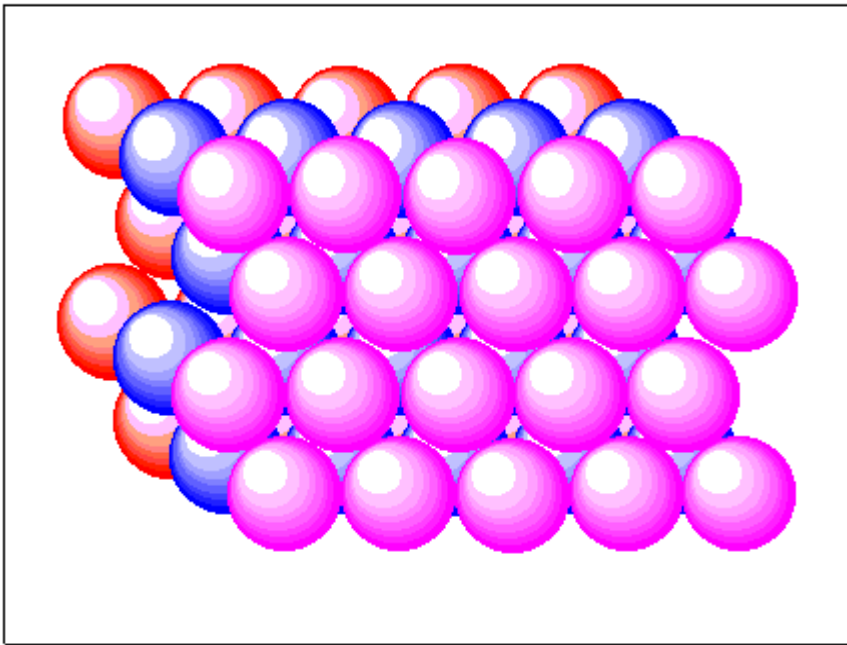
It may occur during

- (1) crystallization from the melt or solid state,
- (2) solid state processes or recrystallization, phase transition, and crystal growth, and
- (3) deformations.



– For FCC metals an error in ABCABC packing sequence

- Intrinsic : Remove a plane (C)
- Extrinsic : Insert an extra plane (A)



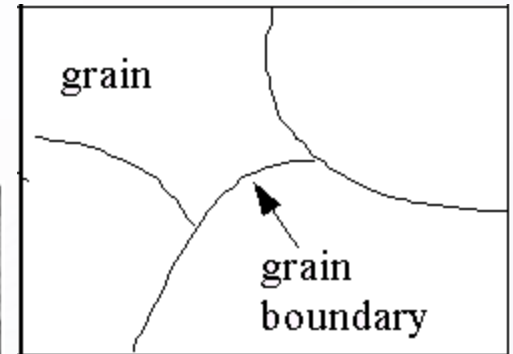
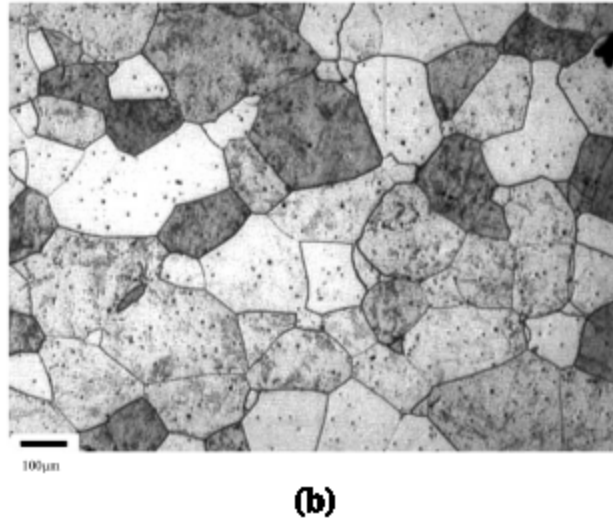
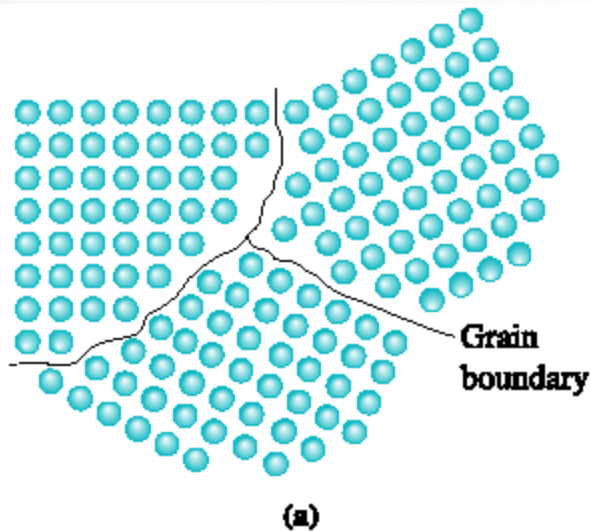
ABCABCABCABC

-
- A 3D model of a crystal lattice structure, showing a regular arrangement of spheres (atoms) in a cubic pattern. The spheres are colored red, blue, and pink, representing different types of atoms or ions.

Diagram illustrating the insertion of a new element into a list. The initial list is A, B, C, A, B, C, A, B, C, A, B, C. A new element 'A' is inserted at the 7th position (under the 6th element 'C'). The resulting list is A, B, C, A, B, C, A, A, B, C, A, B, C.

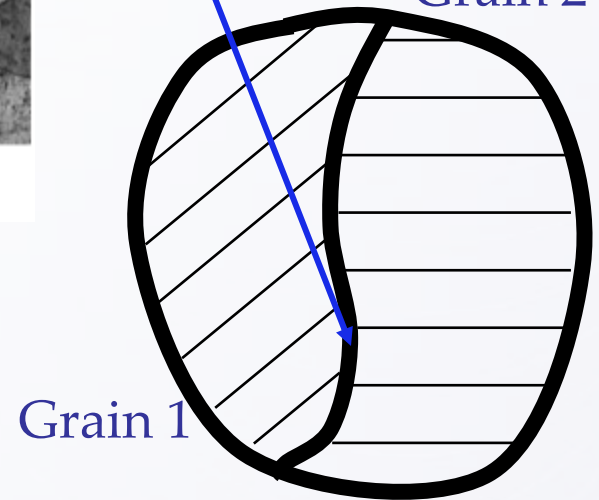


grain boundaries



Grain
Boundary

Grain 2

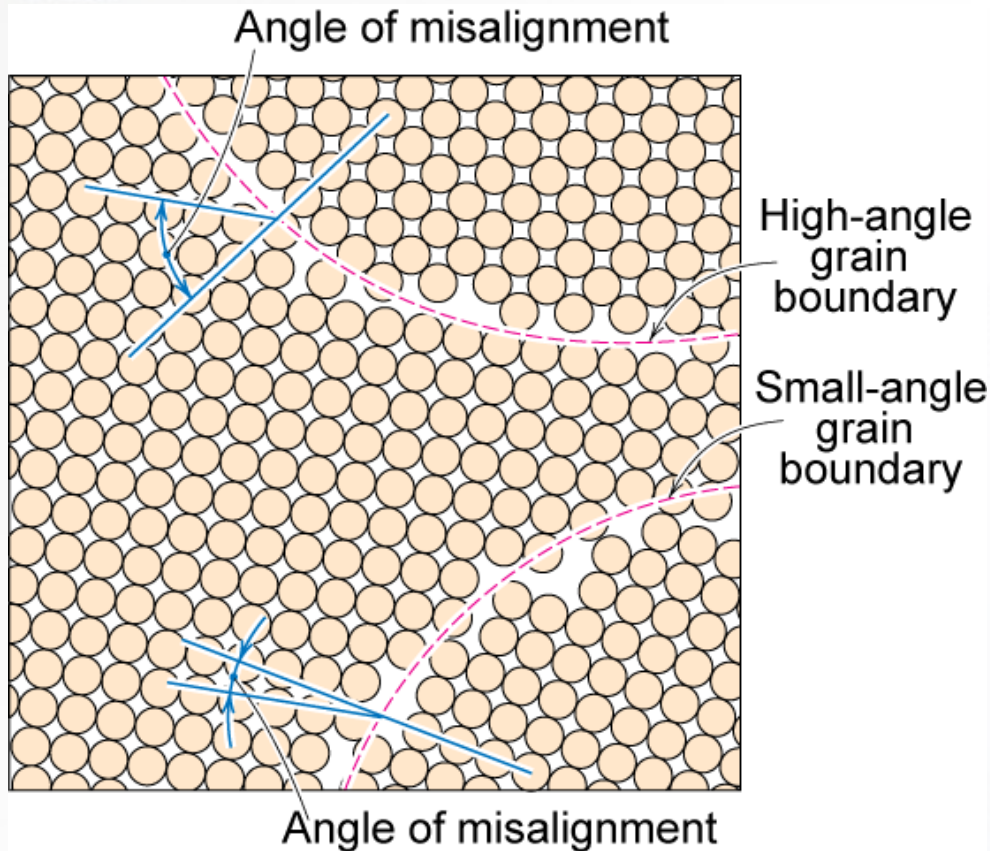


See Figure :

- (a) The atoms near the boundaries of the three grains
- (b) Grains and grain boundaries in a stainless steel sample.

(Courtesy Dr. A. Deardo.)

A grain boundary is a boundary between two regions of identical crystal structure but different orientation



Grains: individual crystals

Grain boundaries: zones between any two grains

- regions between crystals
- transition from lattice of one region to that of the other
- slightly disordered
- low density in grain boundaries
 - high mobility
 - high diffusivity
 - high chemical reactivity



Grain Boundary: *low* and *high* angle

One grain orientation can be obtained by rotation of another grain across the grain boundary about an axis through an **angle**

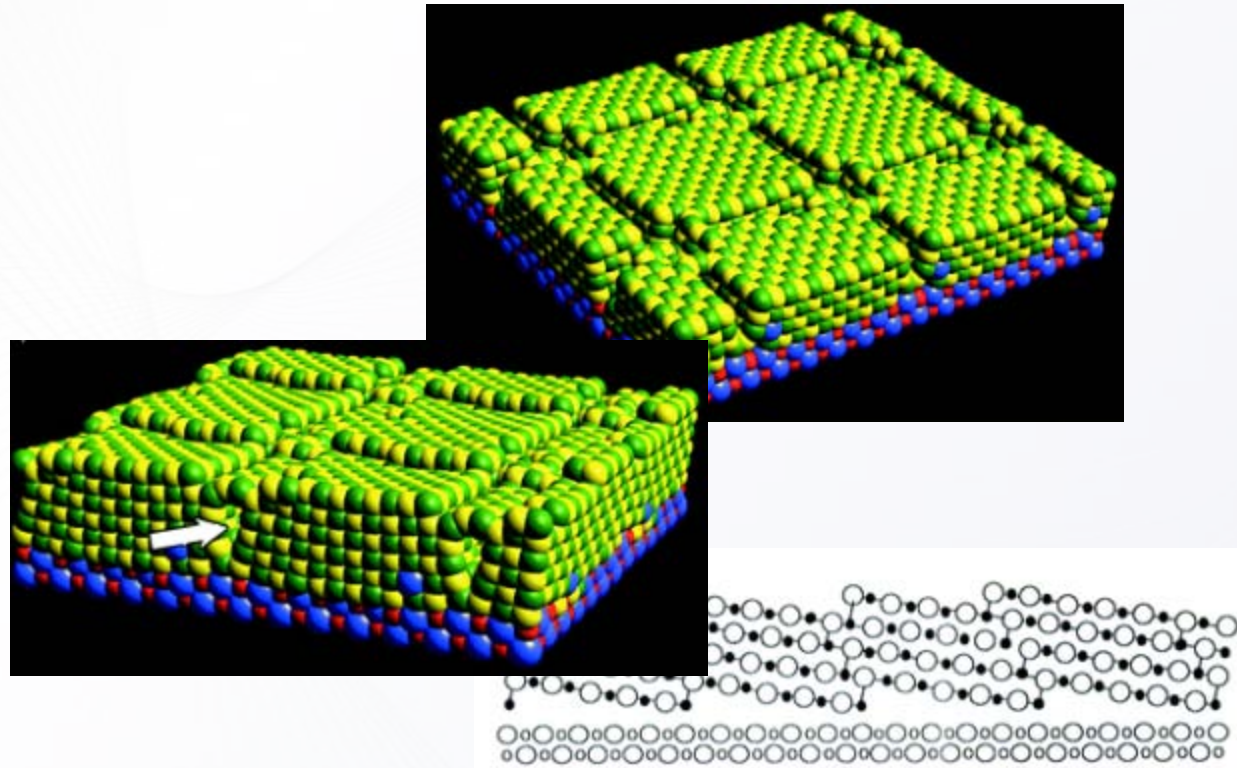
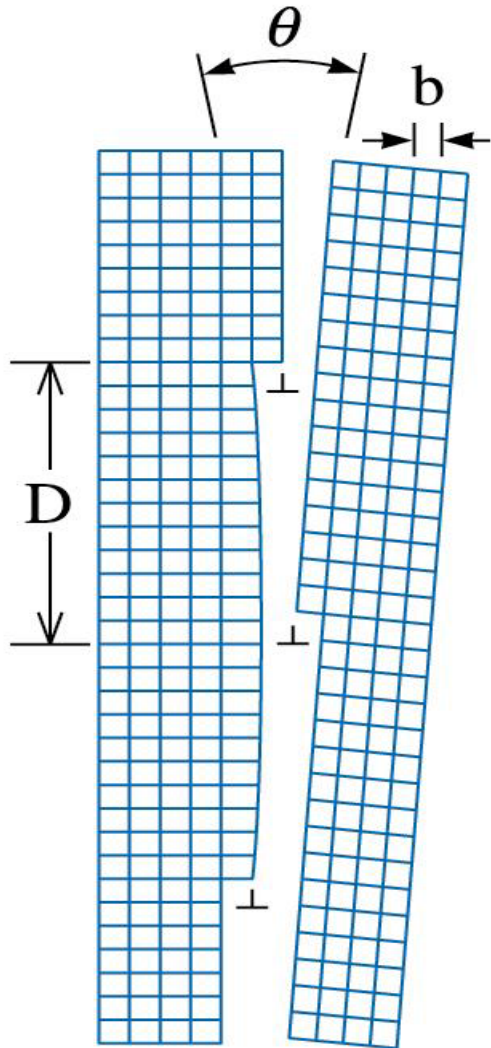
If the angle of rotation is high, it is called a **high angle grain boundary**

If the angle of rotation is low it is called a **low angle grain boundary**

Low-angle grain boundary



An array of dislocations causing a small misorientation of the crystal across the surface of the imperfection.



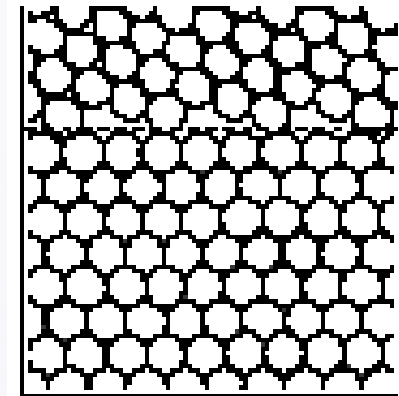
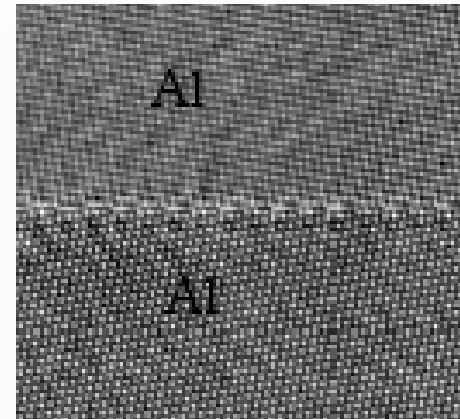
The low angle grain boundary is produced by an array of dislocations, causing an angular mismatch θ between lattices on either side of the boundary.



High-angle grain boundary

High-angle boundaries are likely sites for chemical segregation

A simple high-angle boundary where two crystals meet





Grain Boundary: *tilt* and *twist*

One grain orientation can be obtained by rotation of another grain across the grain boundary about an **axis** through an angle

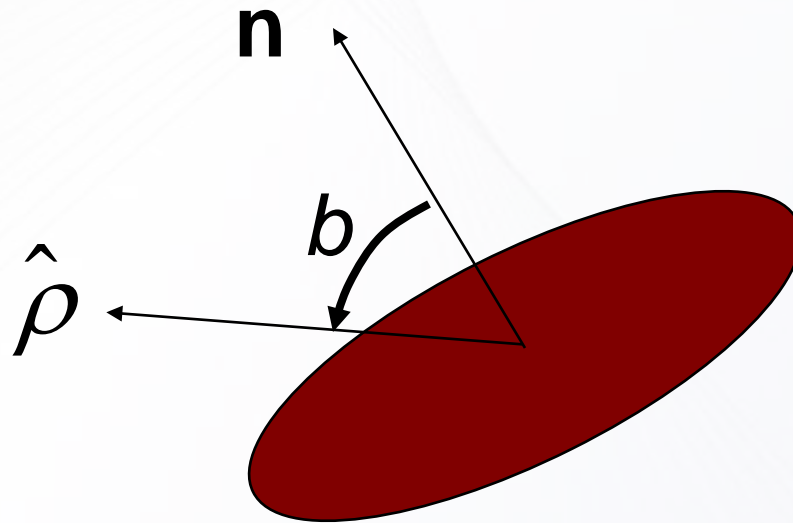
If the axis of rotation lies in the boundary plane it is called **tilt boundary**

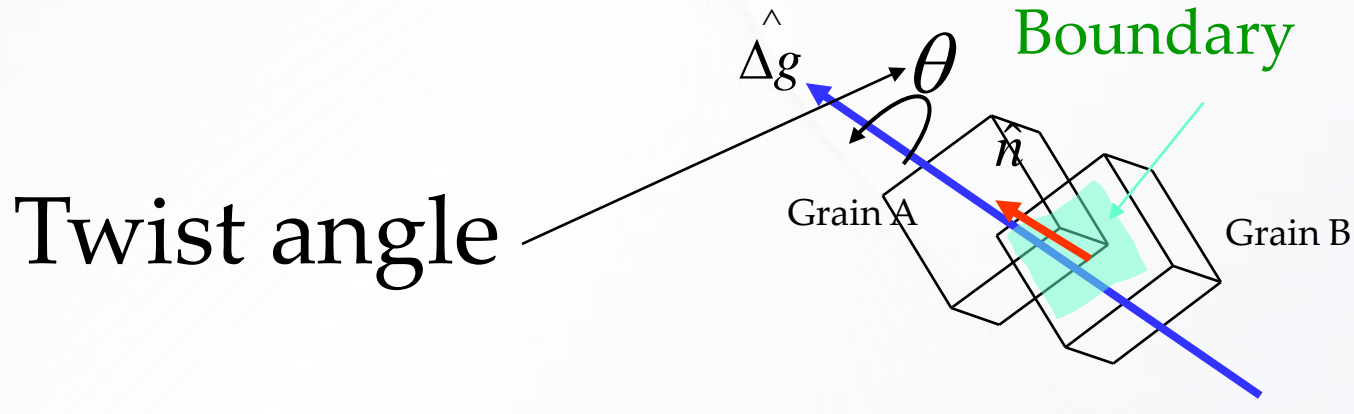
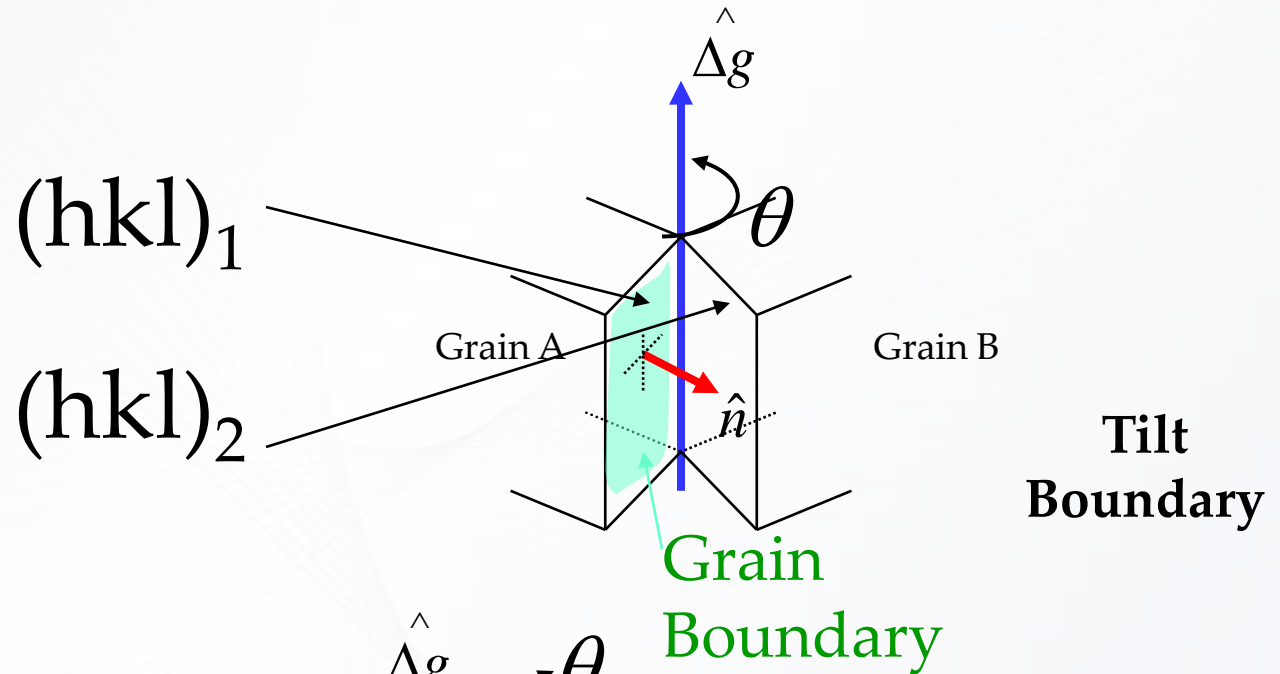
If the angle of rotation is perpendicular to the boundary plane it is called a **twist boundary**

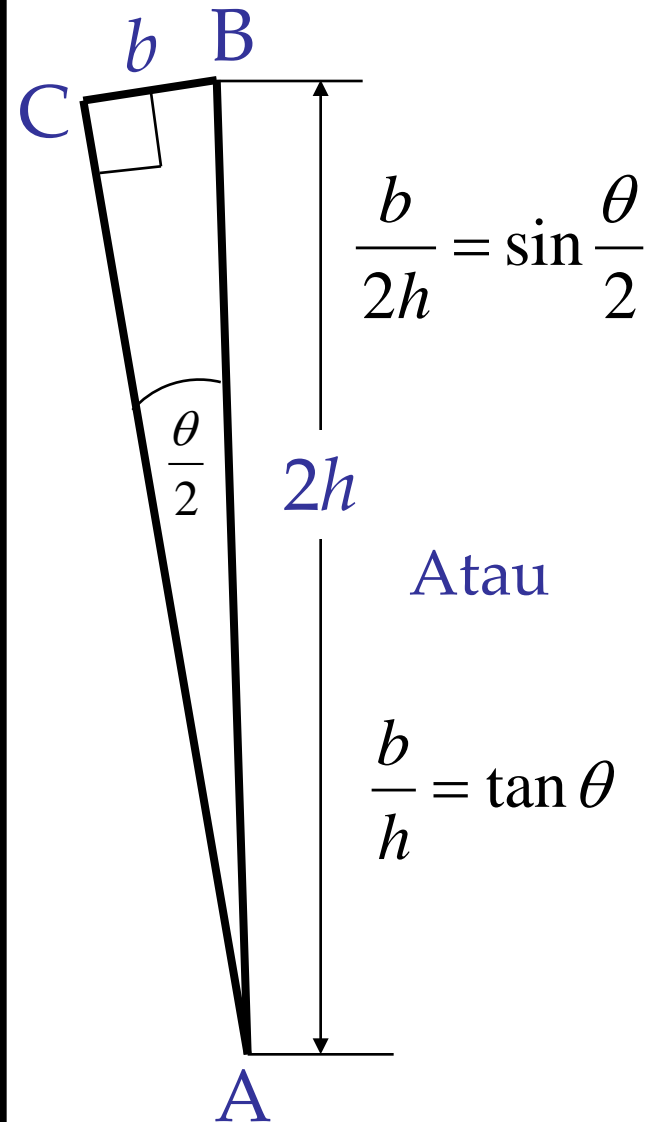
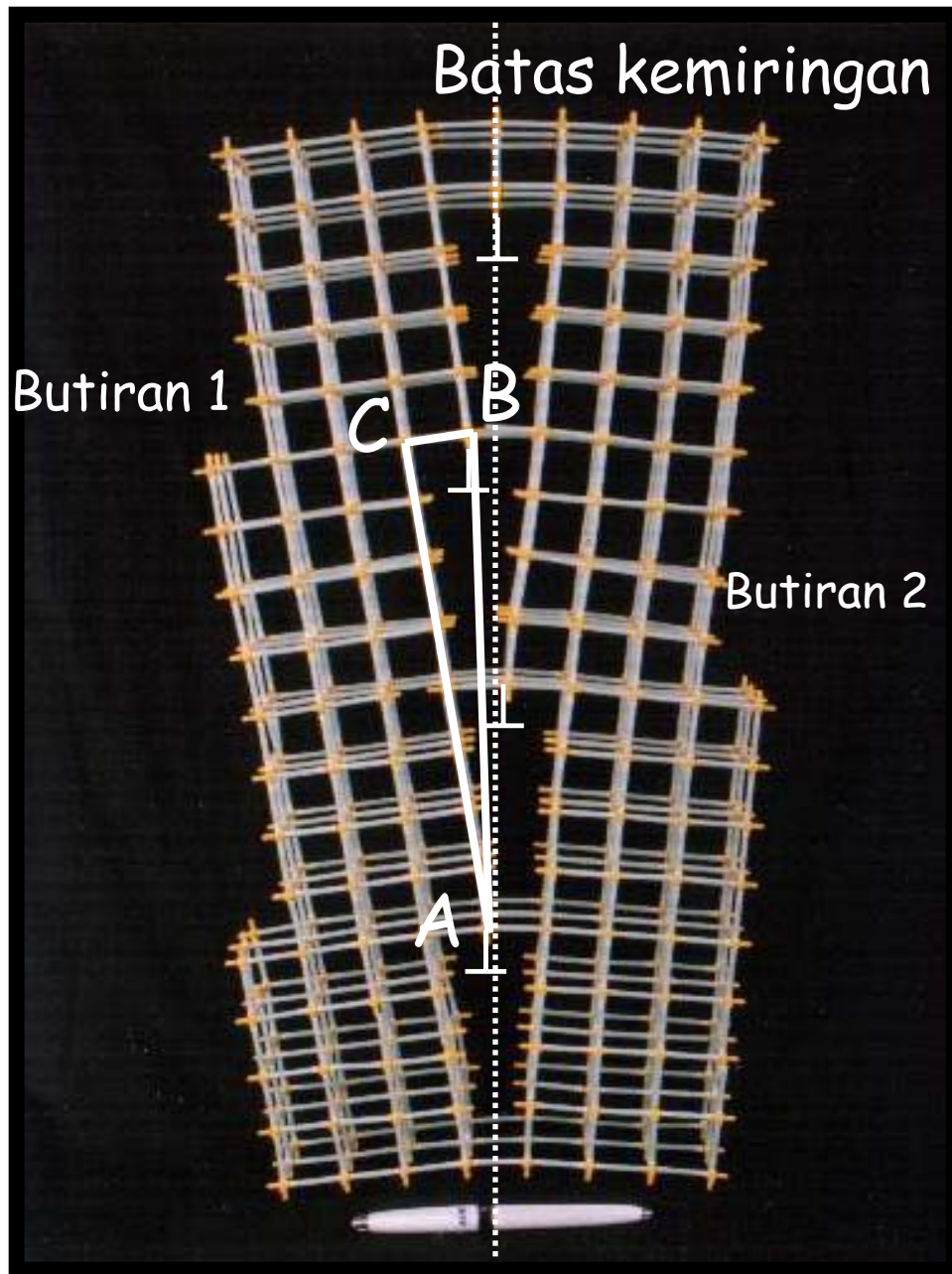
Tilt-twist character

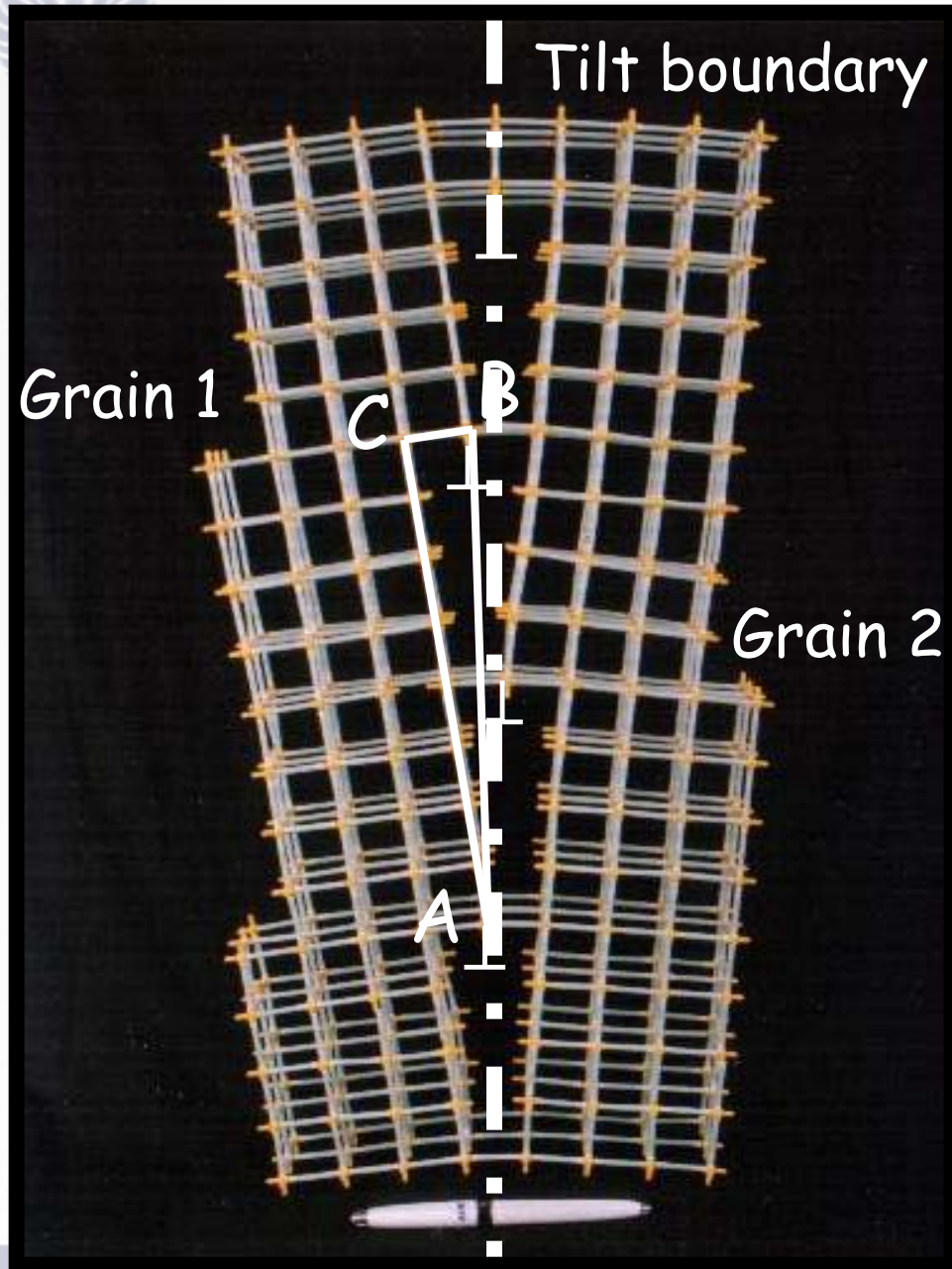
If $\cos^{-1}(b)=0^\circ$, boundary is pure twist;

If $\cos^{-1}(b)=90^\circ$, boundary is pure tilt.







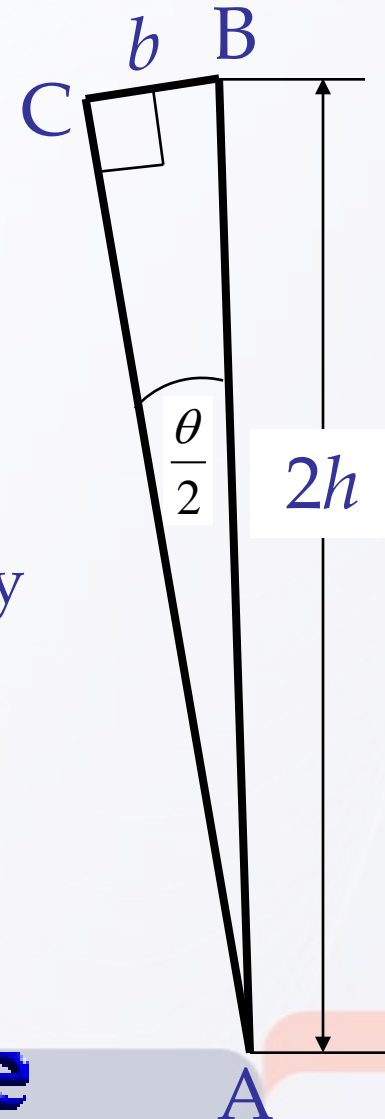


Edge dislocation model
of a small angle tilt
boundary

$$\frac{b}{2h} = \sin \frac{\theta}{2}$$

Or
approximately

$$\frac{b}{h} = \tan \theta$$



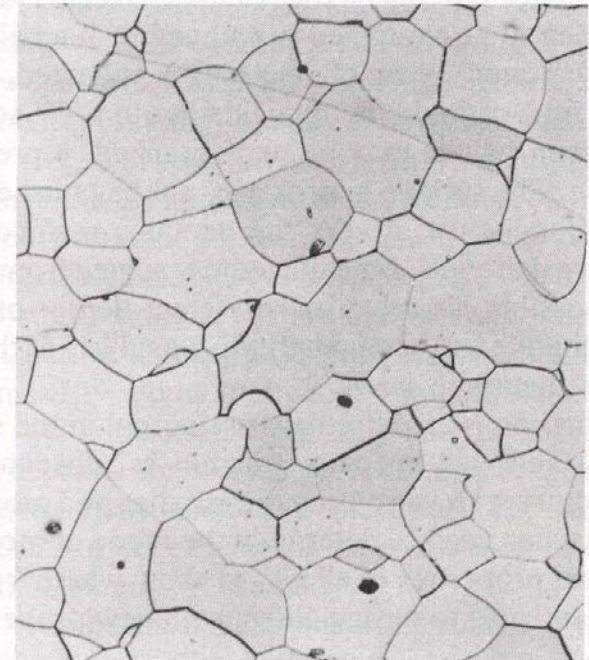
voids (porosity)

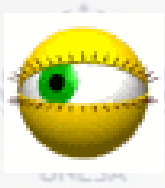
➤ holes in the materials

Voids are small regions where there are no atoms, and can be thought of as clusters of vacancies

inclusions

inclusions particles of foreign matter embedded in the solid





precipitations

Every impurity introduced into a crystal has a certain level of solubility, which defines the concentration of that impurity that the solid solution of the host crystal can accommodate.

Impurity solubility usually decreases with decreasing temperature.



If an impurity is introduced into a crystal at the maximum concentration allowed by its solubility at a high temperature,



the crystal will become supersaturated with that impurity once it is cooled down.



A crystal under such supersaturated conditions seeks and achieves equilibrium by **precipitating** the excess impurity atoms into another phase of different composition or structure.



PRECIPITATES

Impurities cluster together to form small regions of a different phase

Precipitates are considered undesirable because they have been known to act as sites for the generation of dislocations

Precipitates induced during silicon wafer processing come from oxygen, metallic impurities, and dopants like boron



Importance of Defects

- ❑ Effect on Mechanical Properties via Control of the Slip Process
- ❑ Strain Hardening
- ❑ Solid-Solution Strengthening
- ❑ Grain-Size Strengthening
- ❑ Effects on Electrical, Optical, and Magnetic Properties



Defectoscope



Defectoscope

Detect fine surface defects

The system can detect flaws as fine as 30 microns on polished surfaces

Electron microscopy

Optical microscopy